

AlGaN/GaN HEMTs and HBTs

Umesh K. Mishra

PART I

AlGaN/GaN HEMTs

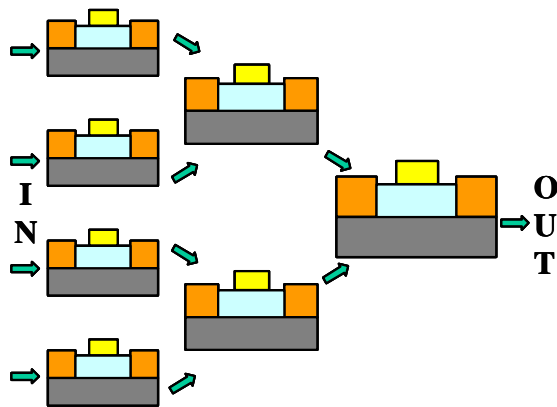
Material	μ	ϵ	E_g	BFOM Ratio	JFM Ratio	Tmax
Si	1300	11.4	1.1	1.0	1.0	300 C
GaAs	5000	13.1	1.4	9.6	3.5	300 C
SiC	260	9.7	2.9	3.1	60	600 C
GaN	1500	9.5	3.4	24.6	80	700 C

BFOM = Baliga's figure of merit for power transistor performance [$K \cdot \mu \cdot E_c^3$]

JFM = Johnson's figure of merit for power transistor performance

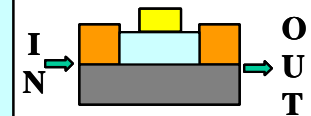
(Breakdown, electron velocity product) [$E_b \cdot V_{br} / 2\pi$]

Need	Enabling Feature	Performance Advantage
High Power/Unit Width	Wide Bandgap, High Field	Compact, Ease of Matching
High Voltage Operation	High Breakdown Field	Eliminate/Reduce Step Down
High Linearity	HEMT Topology	Optimum Band Allocation
High Frequency	High Electron Velocity	Bandwidth, μ -Wave/mm-Wave
High Efficiency	High Operating Voltage	Power Saving, Reduced Cooling
Low Noise	High gain, high velocity	High dynamic range receivers
High Temperature Operation	Wide Bandgap	Rugged, Reliable, Reduced Cooling
Thermal Management	SiC Substrate	High power devices with reduced cooling needs
Technology Leverage	Direct Bandgap: Enabler for Lighting	Driving Force for Technology: Low Cost

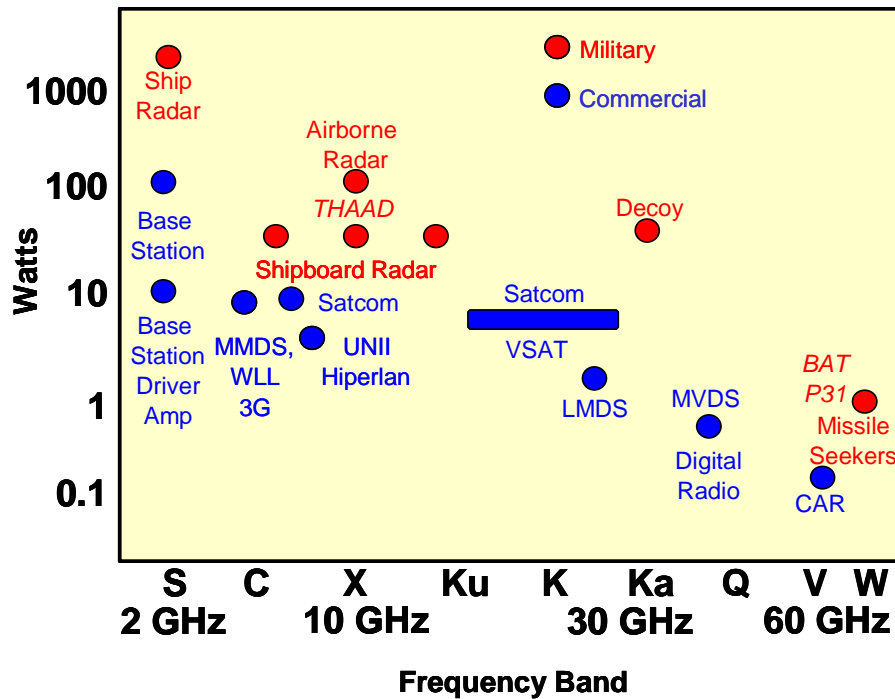


GaAs High Power Amplifier Module

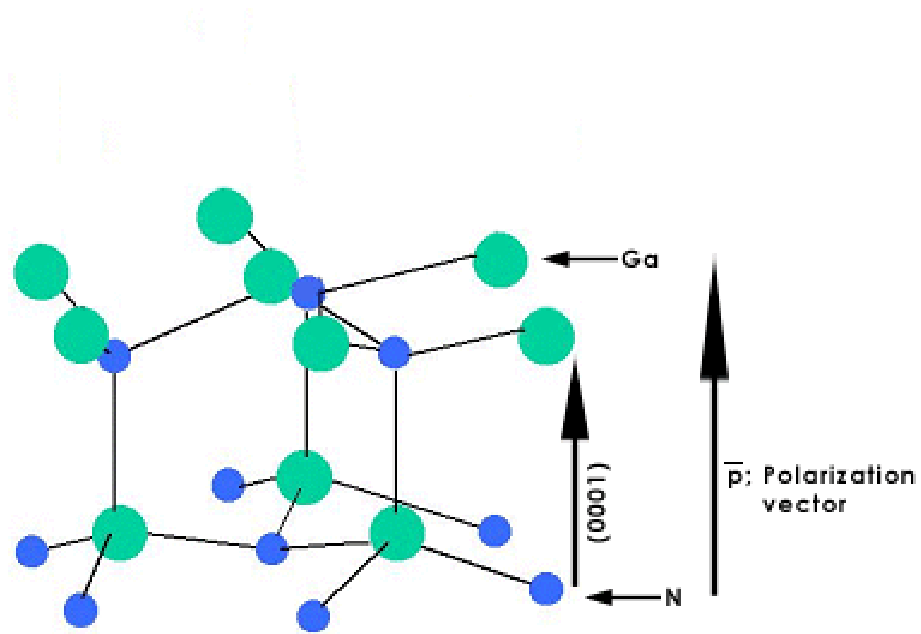
- 10-x power density ($> 10 \text{ W/mm}$)
- 10-x reduction in power-combining
- Improved efficiency ($> 60 \%$)
- Improved reliability
- Compact size
- Superior Performance at reduced cost

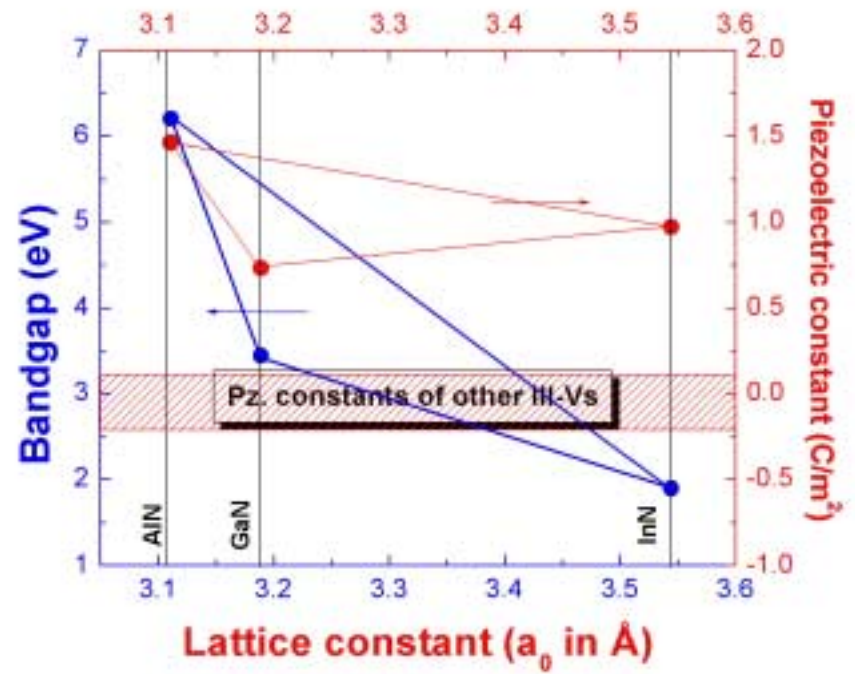
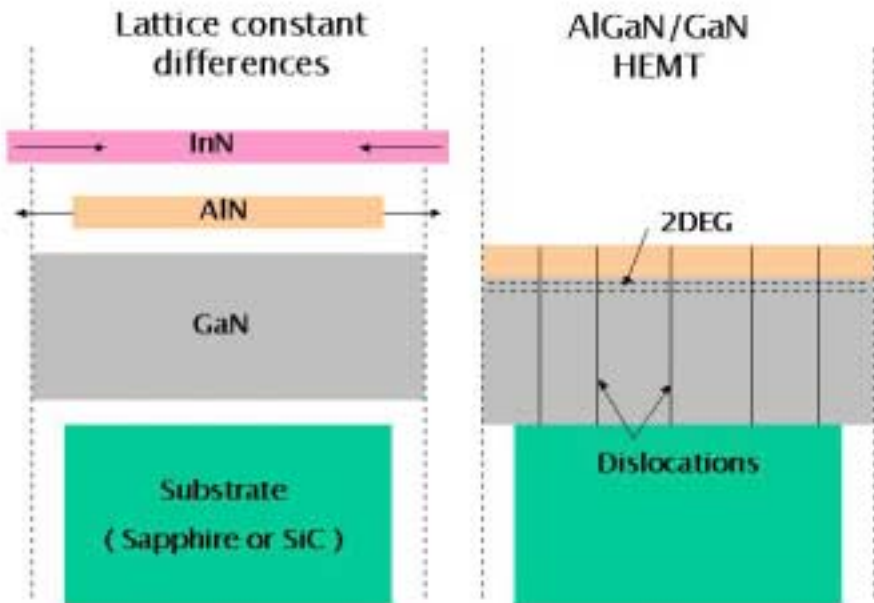


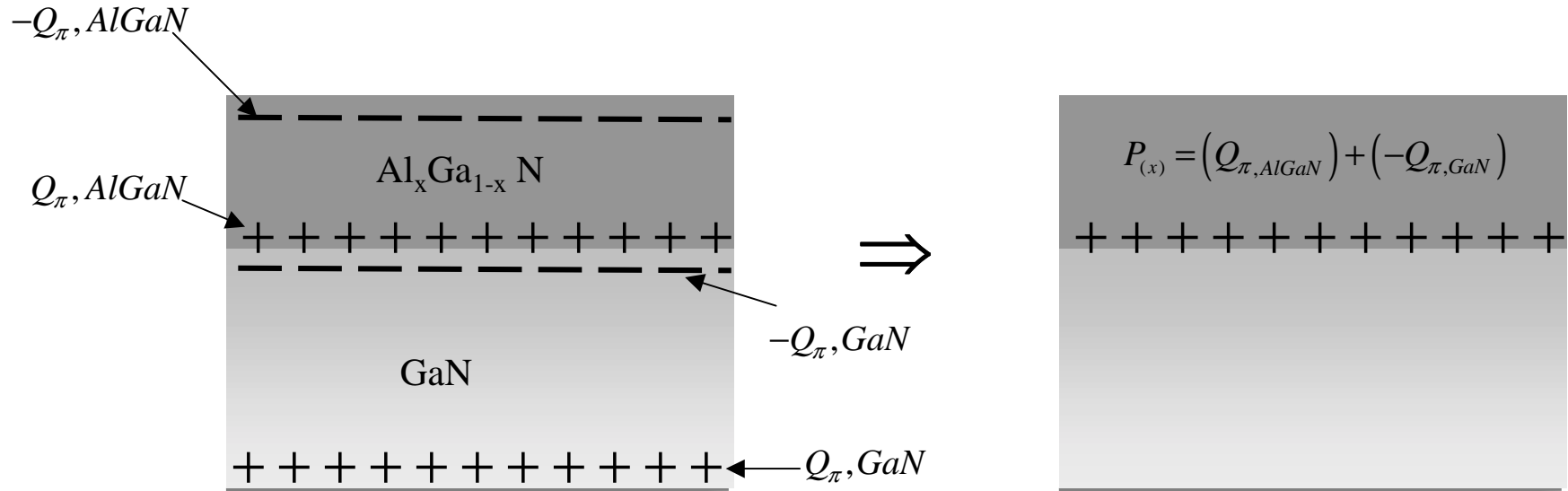
Equivalent High Power GaN Amplifier Module



Ball and Stick Diagram of the GaN Crystal

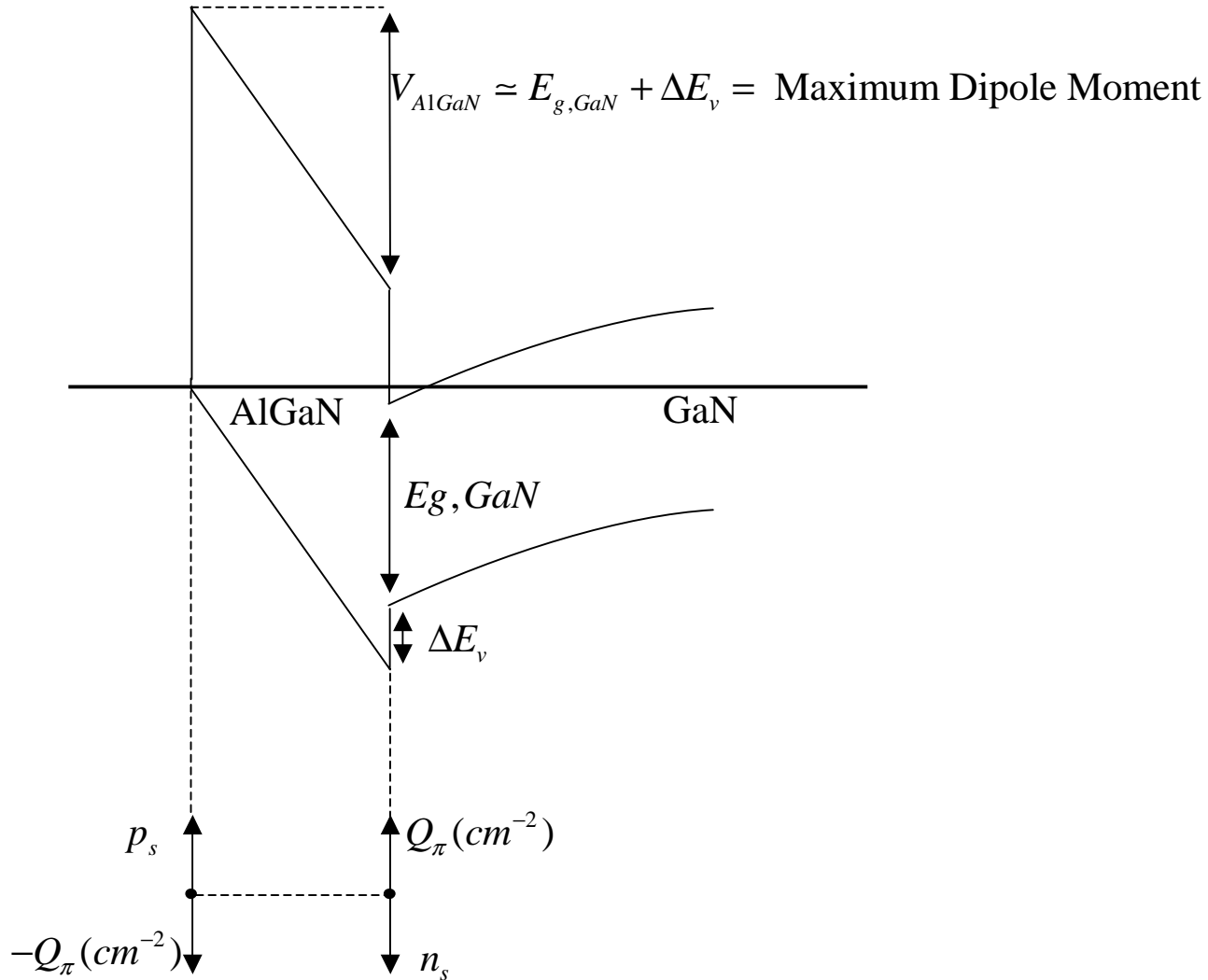






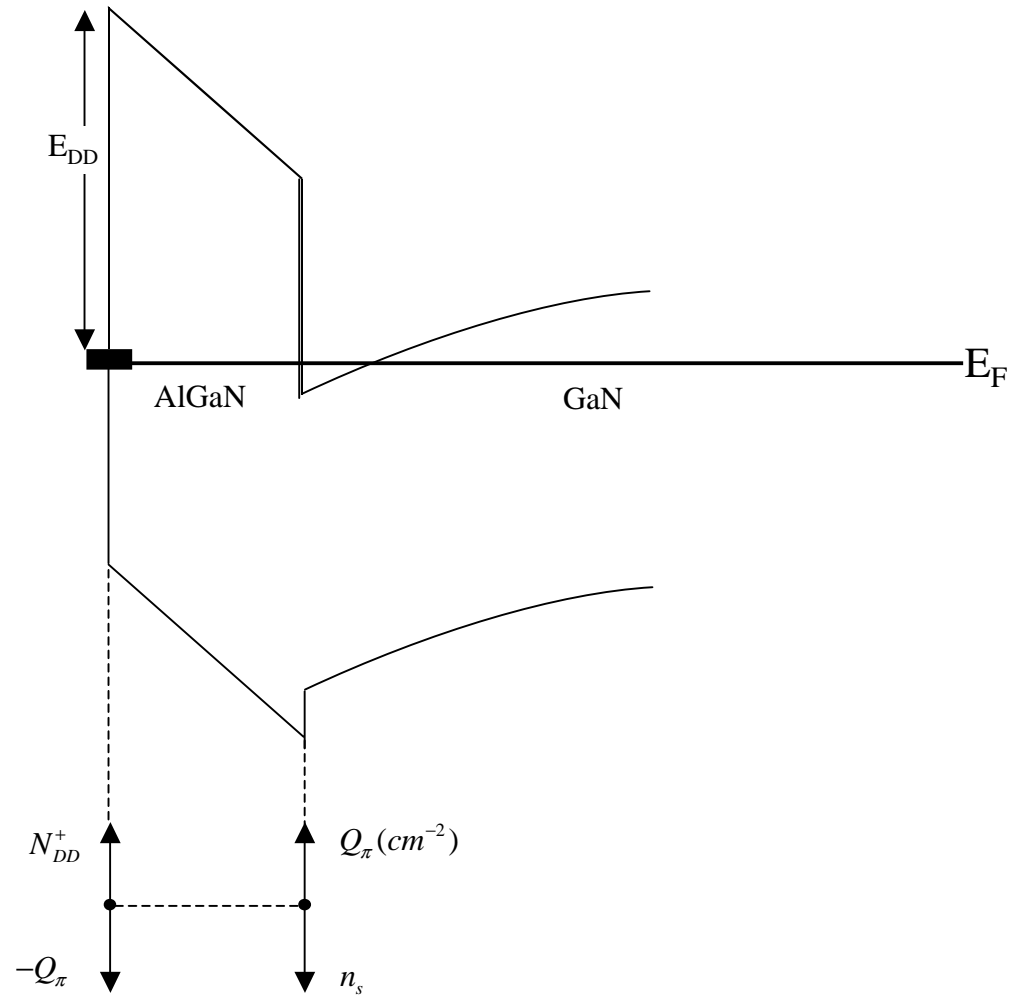
Q_{π} includes the contribution of spontaneous and piezo-electric contributions

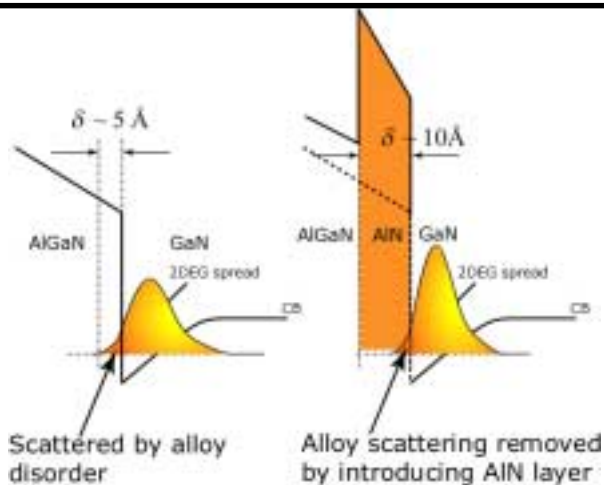
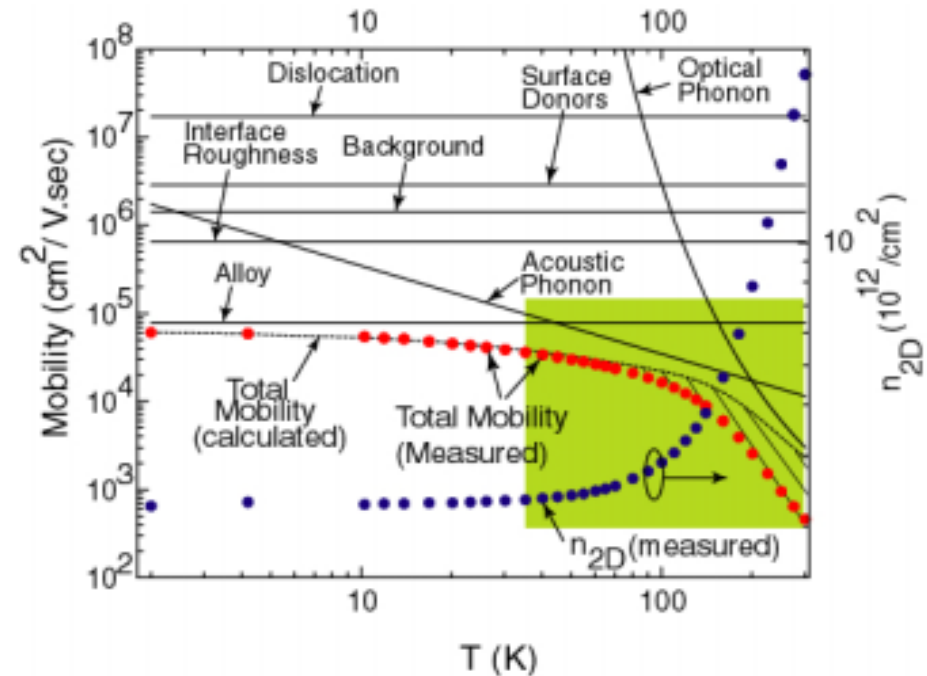
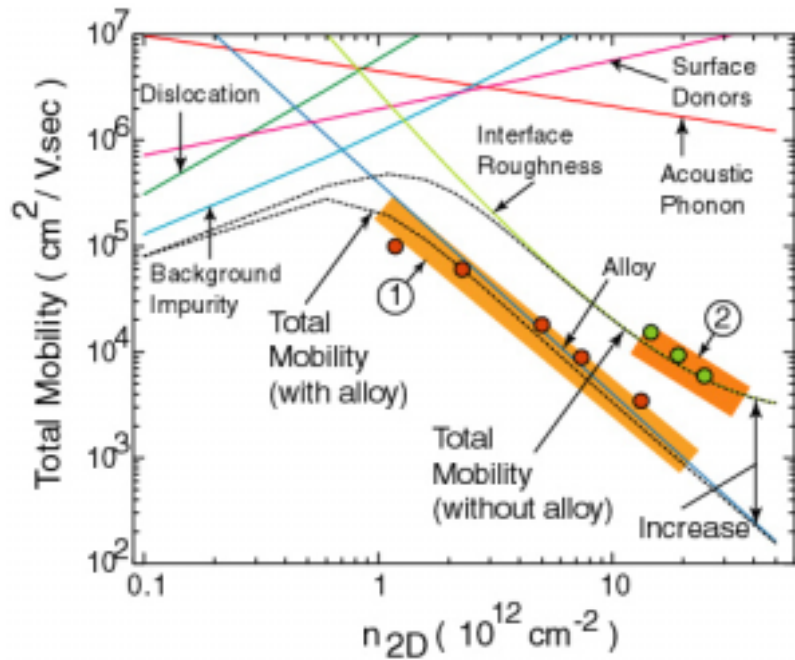
- C



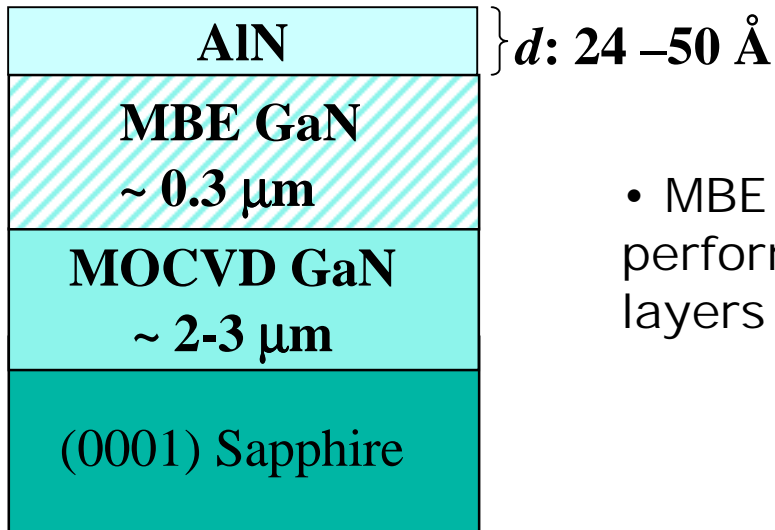
UCSB How does the electron gas form in AlGaIn/GaN structures?

- D



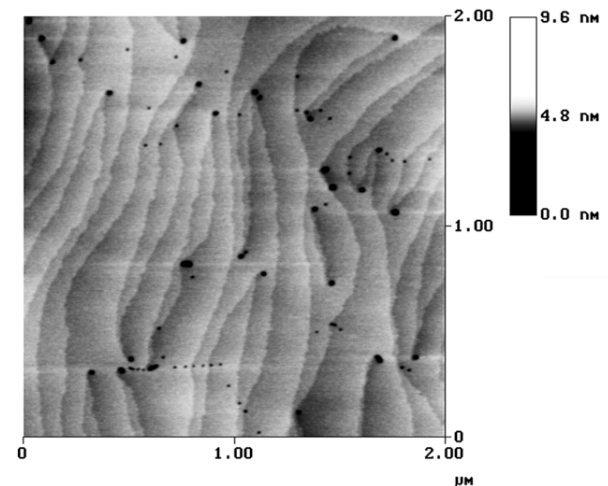


- High density 2DEG mobility limited by alloy scattering.
- Introduce thin AlN interlayer – remove alloy scattering by pushing 2DEG wavefunction out of the alloy region.
- Very useful result for attaining high conductivity 2DEG channels for HEMTs.

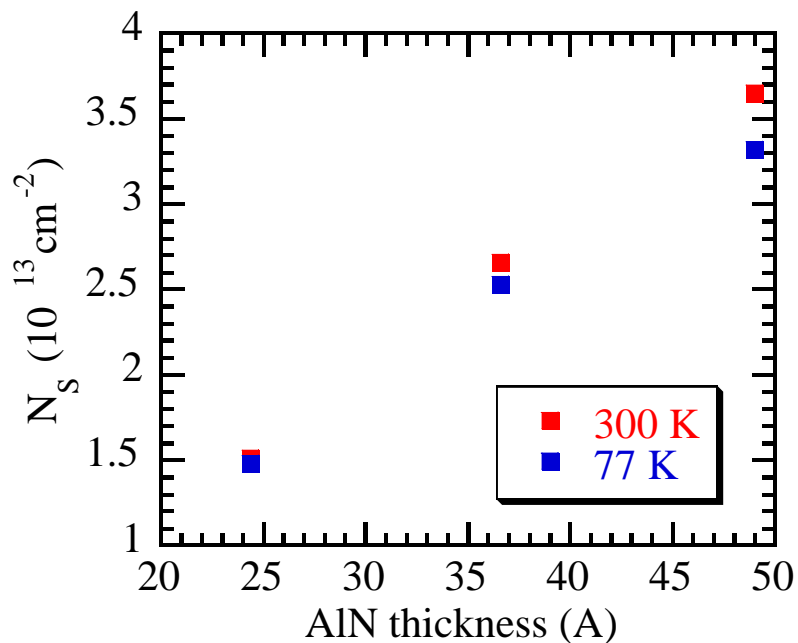


- MBE growth of AlN/GaN structures is performed on GaN “templates” – thick GaN layers grown by MOCVD on (0001) sapphire

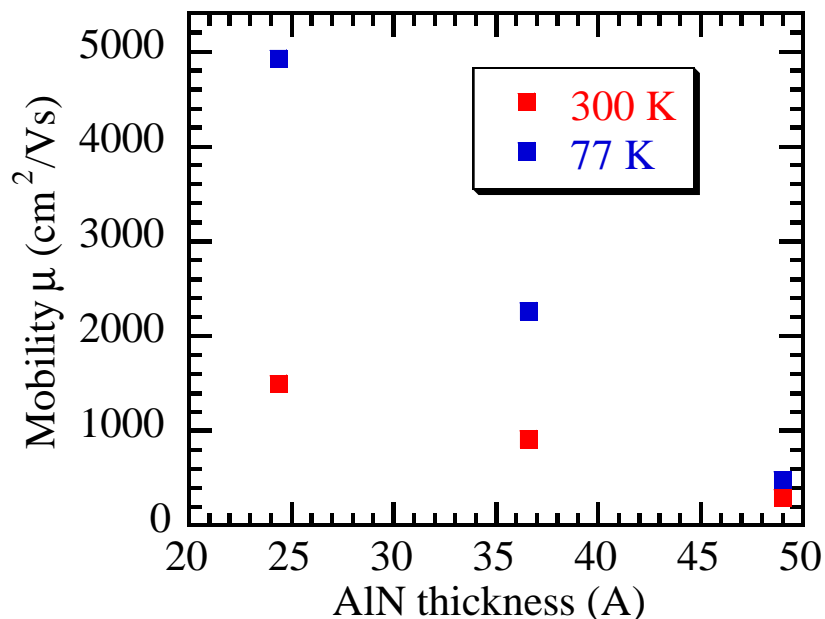
- Templates of two types are employed:
 - (a) unintentionally doped GaN
(dislocation density: $\sim 5 \times 10^8 - 10^9 \text{ cm}^{-2}$)
 - (b) semi-insulating GaN
(dislocation density: $\sim 10^{10} \text{ cm}^{-2}$)
- Ga-stable growth (III/V flux ratio > 1)
- $T_s \sim 740 \text{ }^\circ\text{C}$



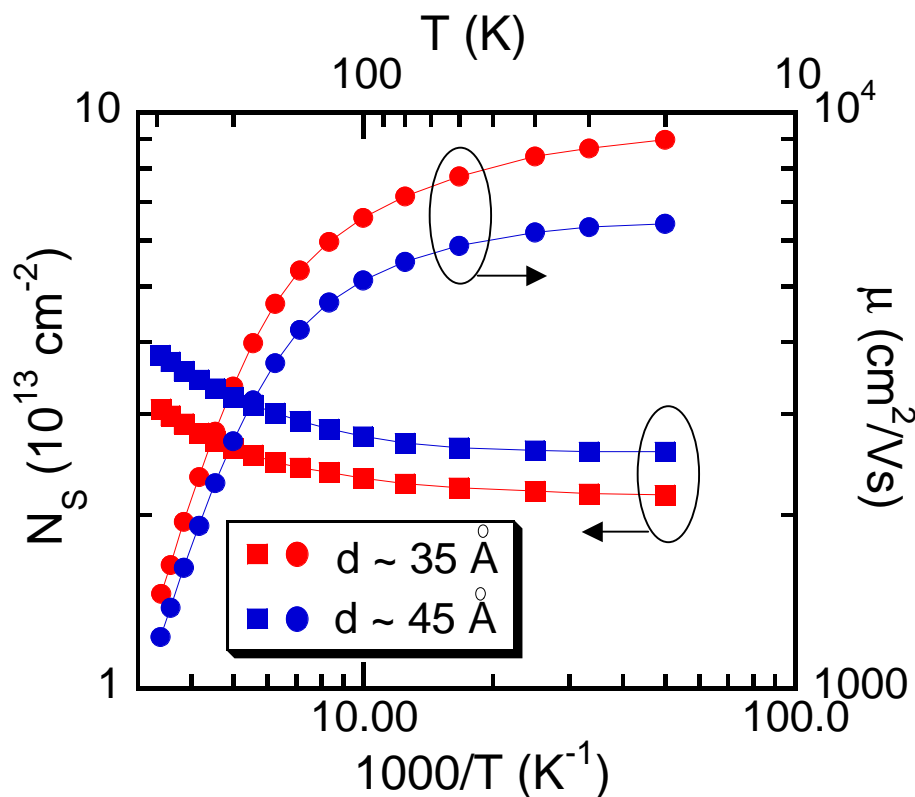
AFM image of type (a)
GaN template



- 2DEG sheet density reaches the value of $3.65 \times 10^{13} \text{ cm}^{-2}$ in the AlN/GaN structure with a 49 \AA barrier



- Both room-temperature and 77 K electron mobility decrease drastically as AlN barrier width increases



AlN ~ 35 Å

$$\mu(300 \text{ K}) = 1460 \text{ cm}^2/\text{Vs}$$

$$N_{2\text{DEG}} \approx N_s(20 \text{ K}) = 2.2 \times 10^{13} \text{ cm}^{-2}$$

AlN ~ 45 Å

$$\mu(300 \text{ K}) = 1230 \text{ cm}^2/\text{Vs}$$

$$N_{2\text{DEG}} \approx N_s(20 \text{ K}) = 2.7 \times 10^{13} \text{ cm}^{-2}$$

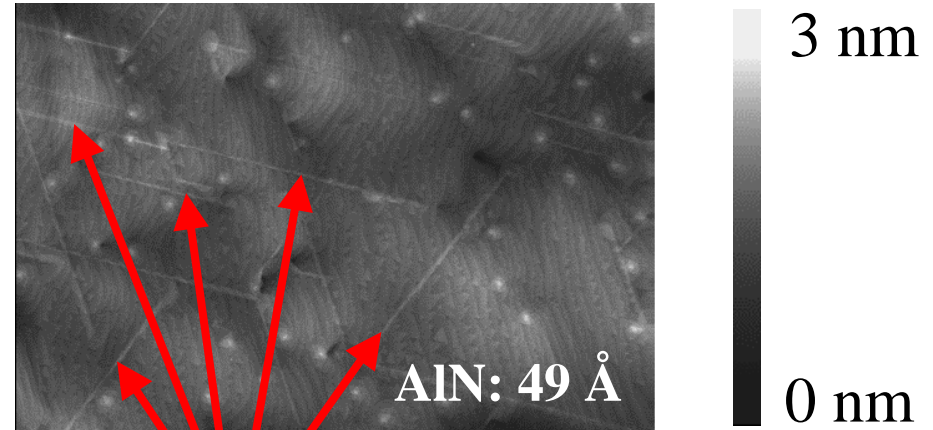
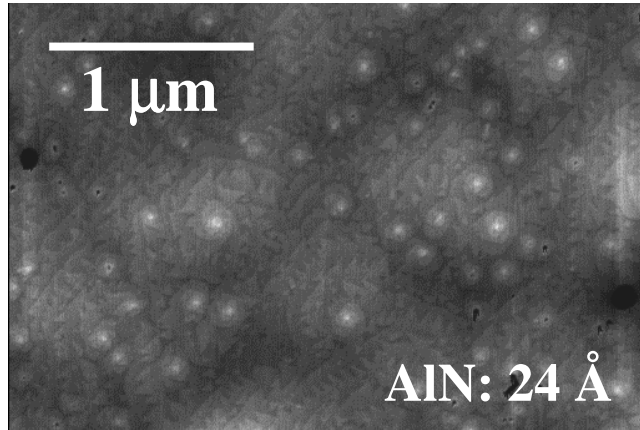
Pessimistic estimate of the
2DEG sheet resistance at 300 K:

$$R_{\square} < 200 \ \Omega/\square$$

AlN ~ 50 Å

$$\mu(300 \text{ K}) = 330 \text{ cm}^2/\text{Vs}; N_s(300 \text{ K}) = 5.6 \times 10^{13} \text{ cm}^{-2}$$

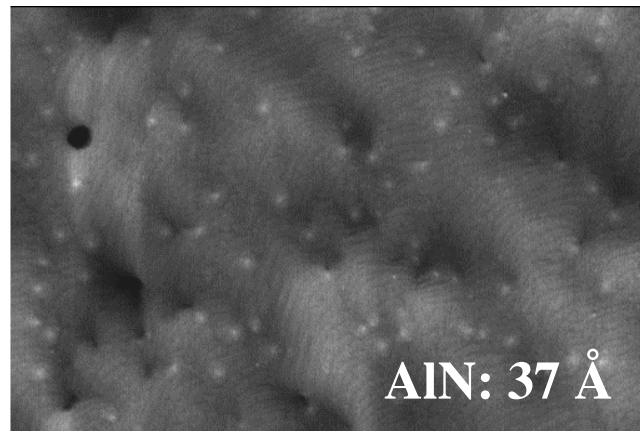
$$\mu(77 \text{ K}) = 660 \text{ cm}^2/\text{Vs}; N_s(77 \text{ K}) = 3.6 \times 10^{13} \text{ cm}^{-2}$$

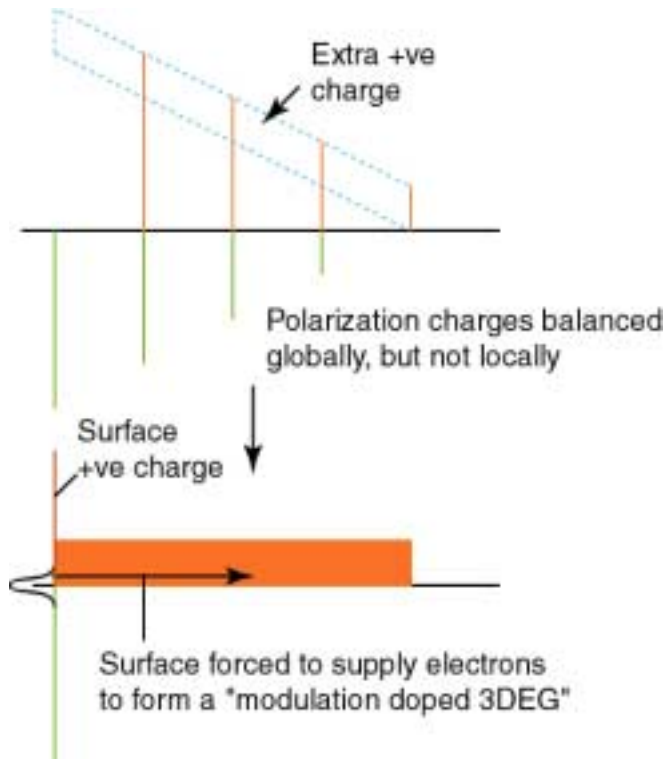
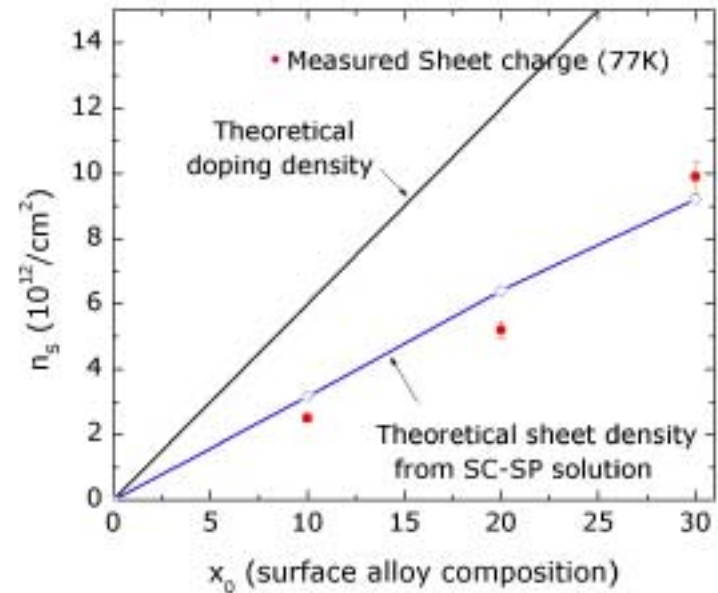
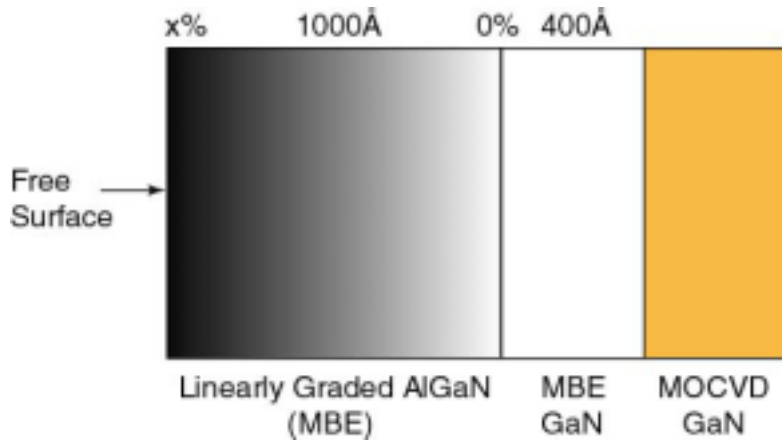


Cracks in AlN layer

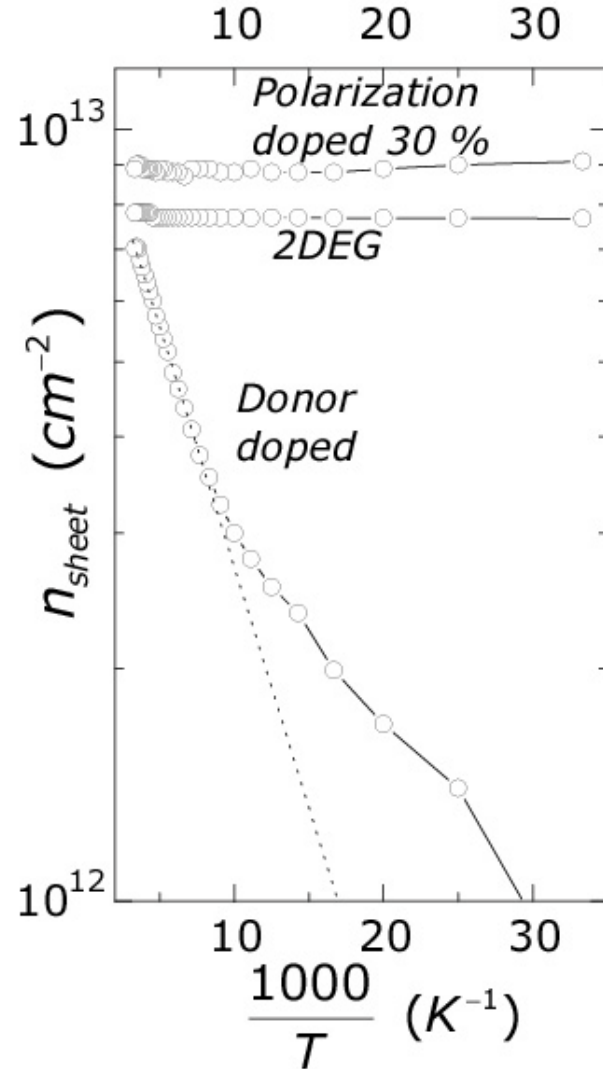
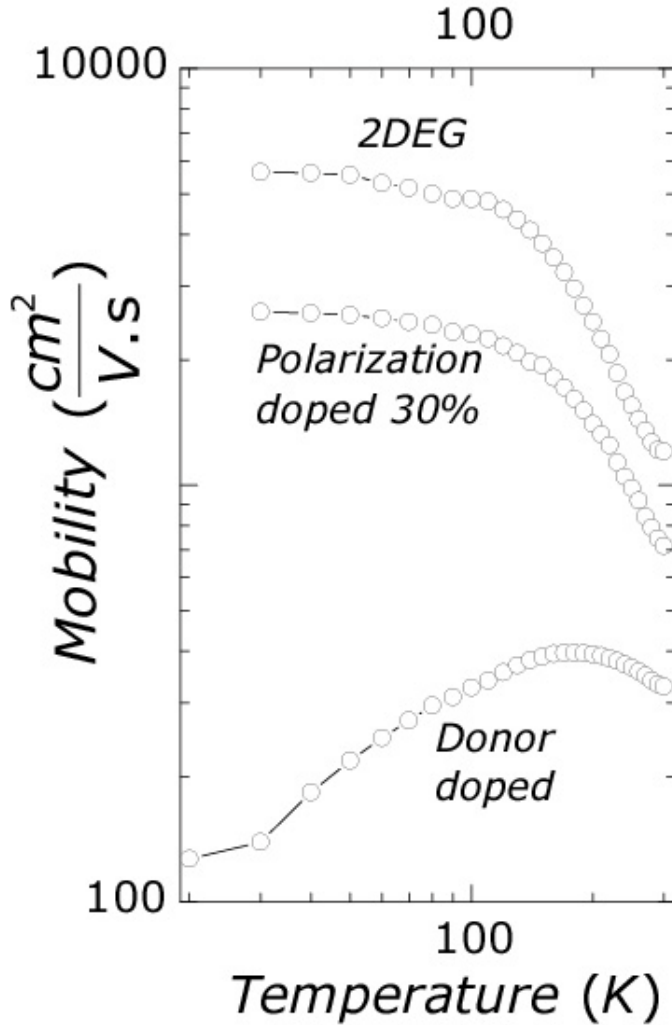


Tensile strain
relaxation process
begins at $d \sim 49 \text{ \AA}$

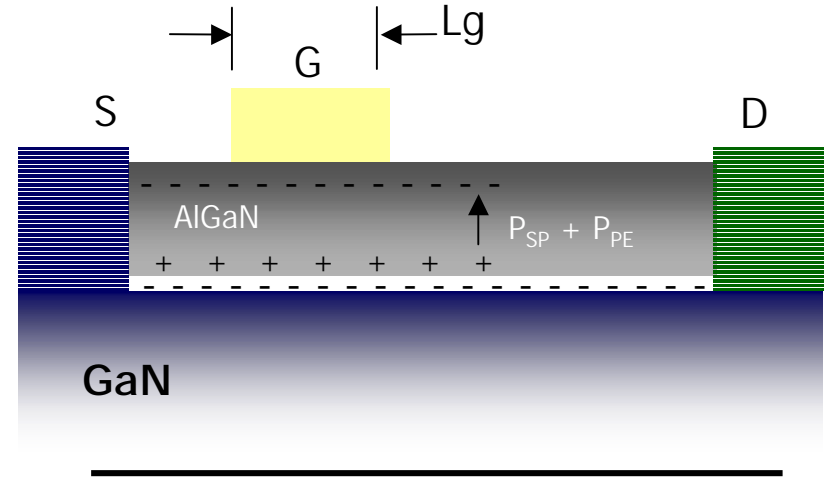
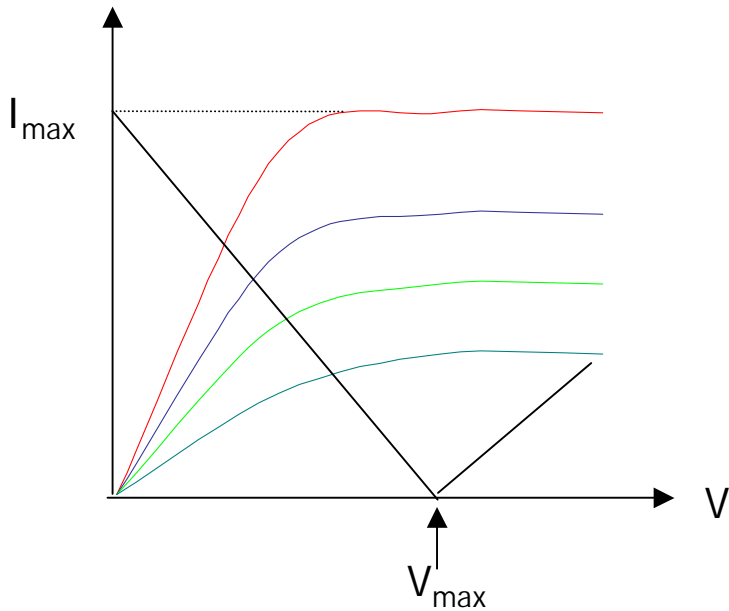




- 3D electron slab by Polarization doping demonstrated.
- Carrier density verified by self-consistent Schrodinger – Poisson calculations.
- How do transport properties of the 3DES compare to the donor doped and 2DEG counterparts ?



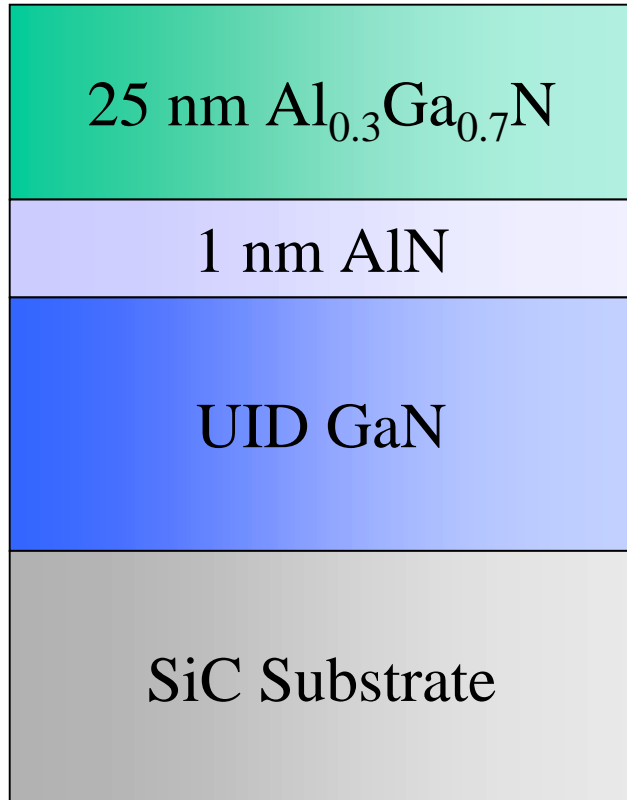
If it ain't good @ DC it ain't goin' to be good @ RF



$$P_{\max} = \frac{1}{8} V_{\max} \cdot I_{\max}$$

$$I = V \cdot n_S \cdot v$$

- Maximize I \Rightarrow Maximize n_S, v
- Maximize n_S \Rightarrow Maximize P_{SP}, P_{PE}
 \Rightarrow Maximize Al mole fraction without strain relaxation
- Maximize v \Rightarrow Minimize effective gate length
 \Rightarrow Minimize L_g and gate length extension
- Maximize μ \Rightarrow Minimize dislocations
 \Rightarrow Smooth interface

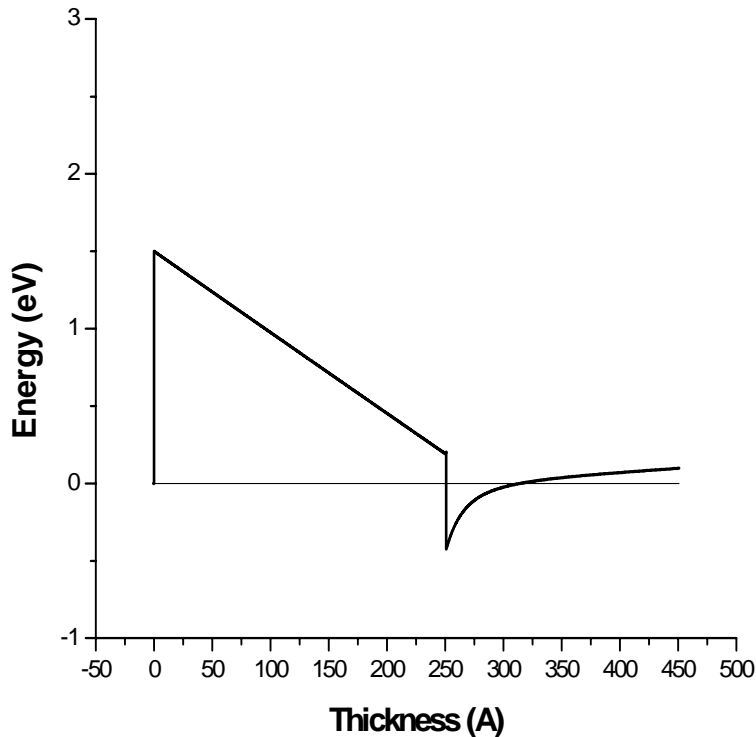


+ High charge

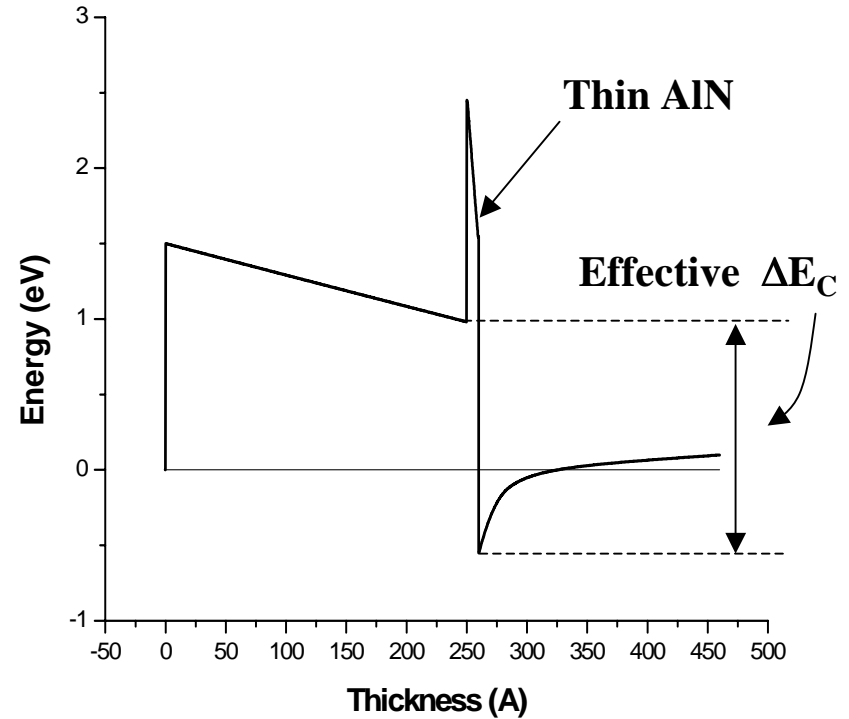
+ High mobility

Hall Data:

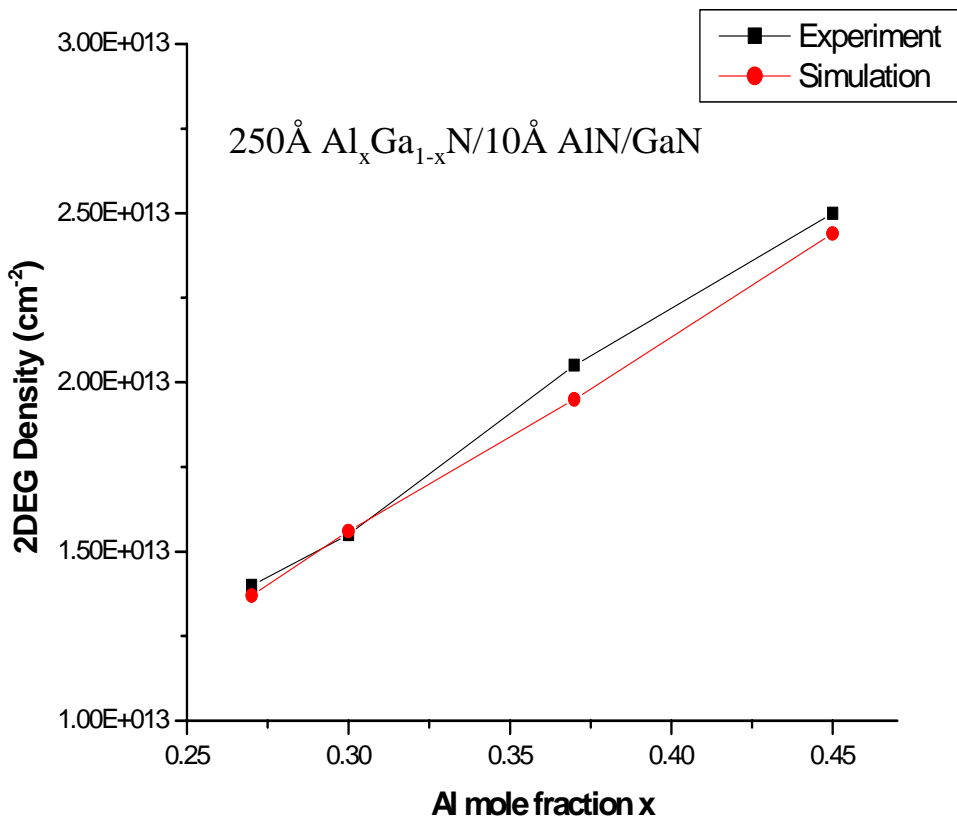
- Conventional $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}/\text{GaN}$ HEMT
 $n_s = 1.2 \times 10^{13} \text{ cm}^{-2}$
 $\mu = 1200 \text{ cm}^2/\text{V/s}$
- $\text{Al}_{0.3}\text{Ga}_{0.7}\text{N}/\text{AlN}/\text{GaN}$ HEMT
 $n_s = 1.65 \times 10^{13} \text{ cm}^{-2}$
 $\mu = 1716 \text{ cm}^2/\text{V/s}$

250 Å Al_{0.3}GaN/GaN HEMT

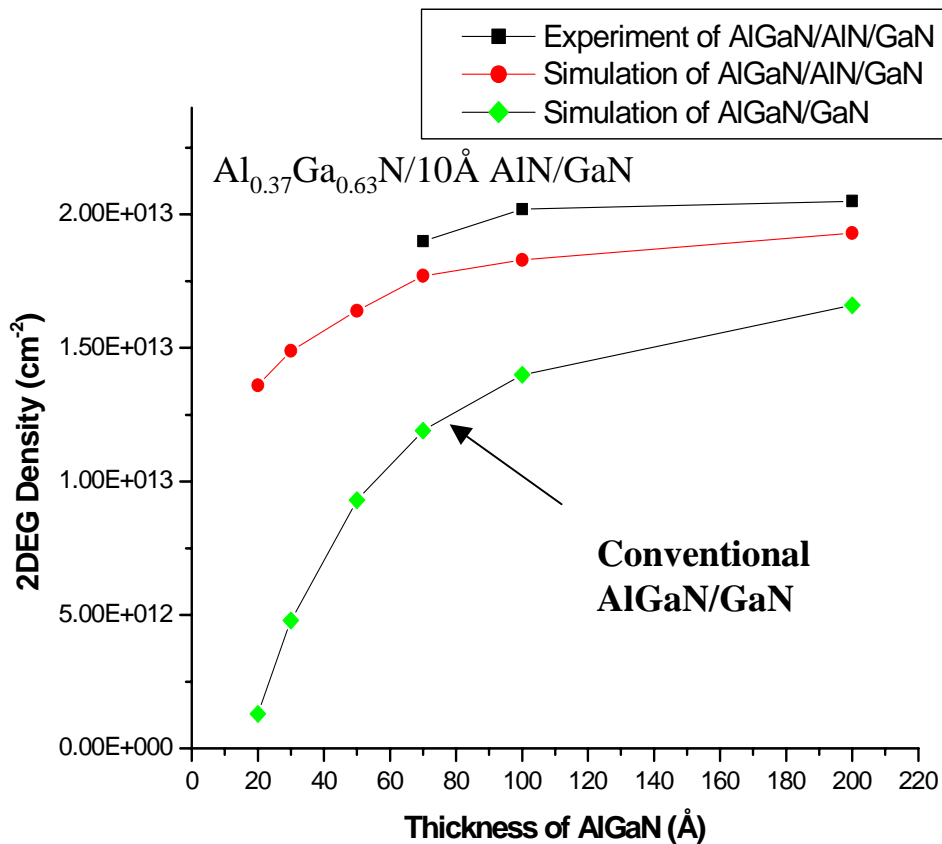
- $n_s = 1.35 \times 10^{13} \text{ cm}^{-2}$

250 Å Al_{0.3}GaN/ 10 Å AlN/GaN HEMT

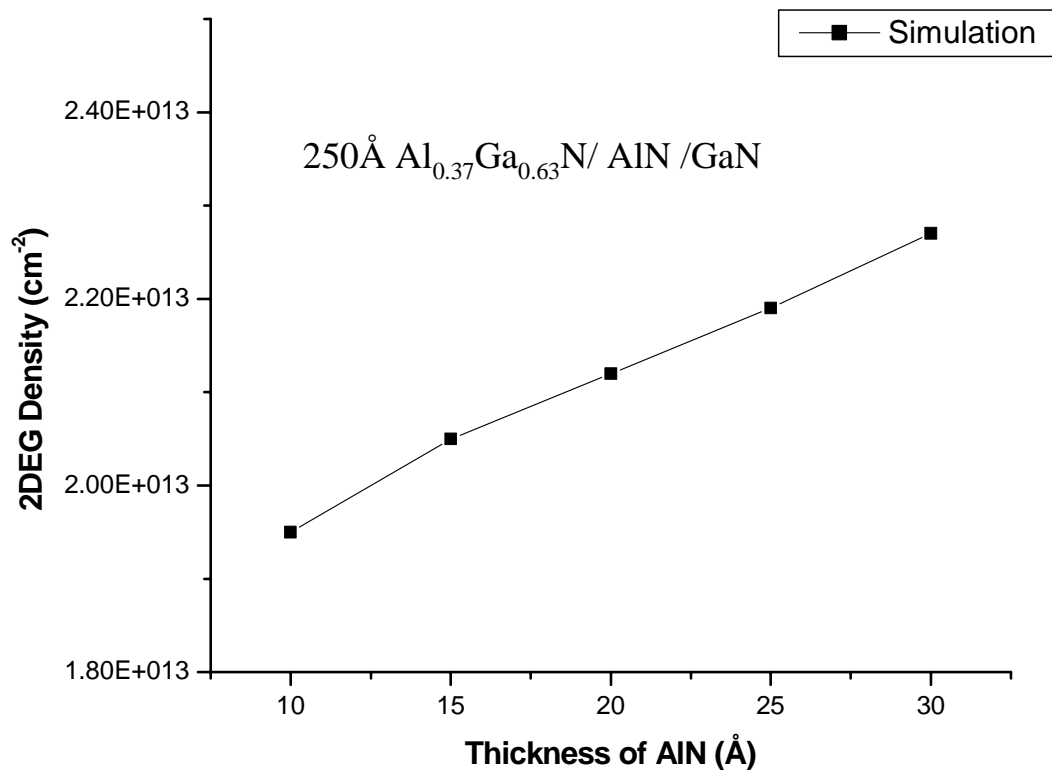
- Higher charge: $n_s = 1.56 \times 10^{13} \text{ cm}^{-2}$ due to higher effective ΔE_C
- Higher mobility due to the removal of alloy disorder scattering



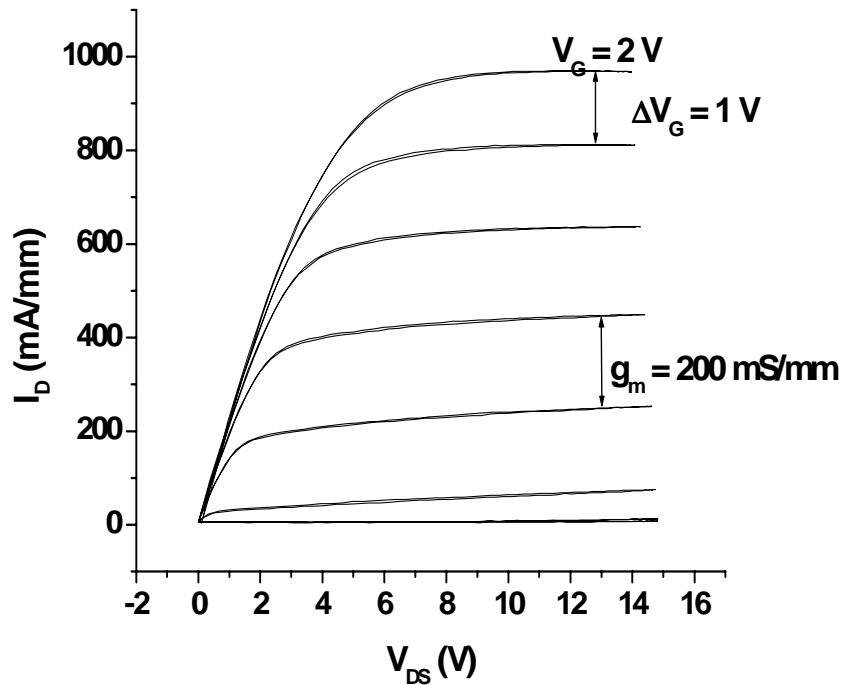
- Strongly dependent on the Al mole fraction



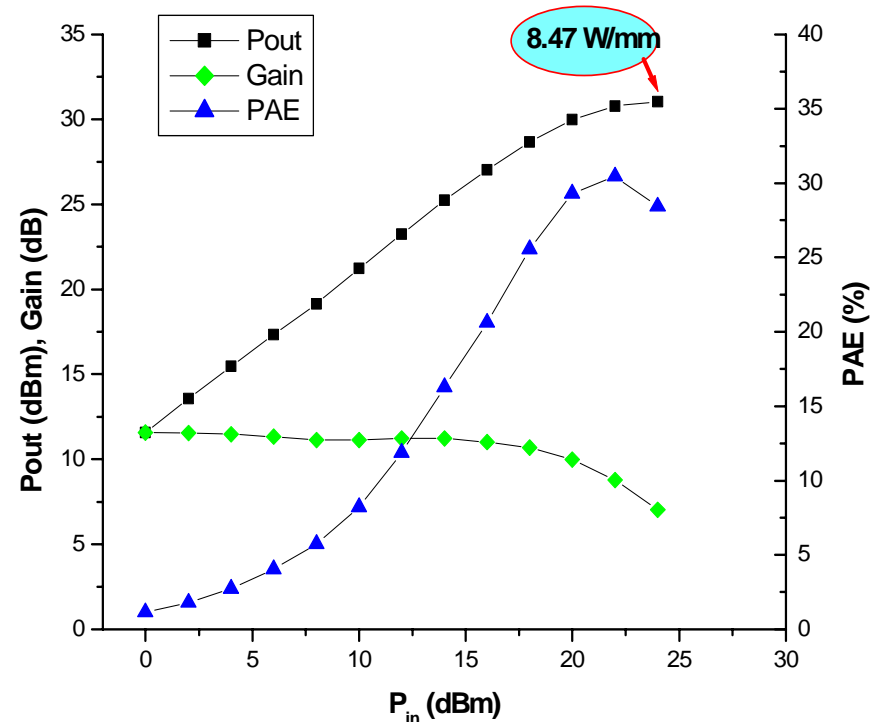
- Weak function of AlGaIn thickness
- Faster saturation than conventional AlGaIn/GaN HEMT



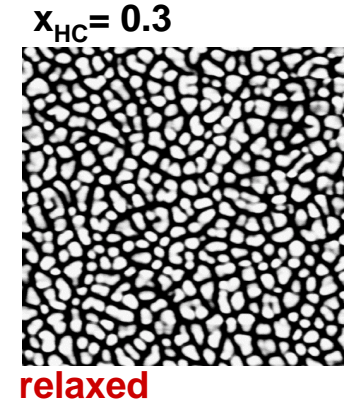
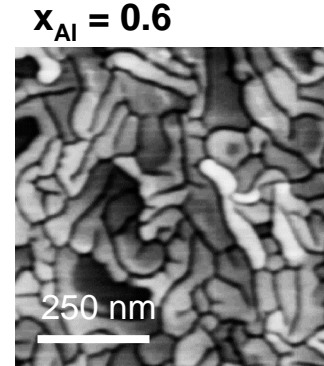
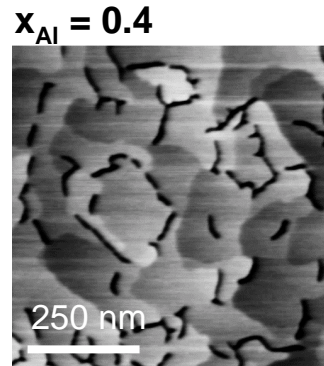
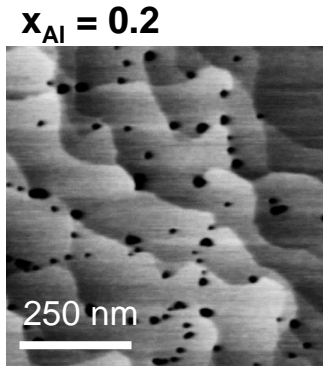
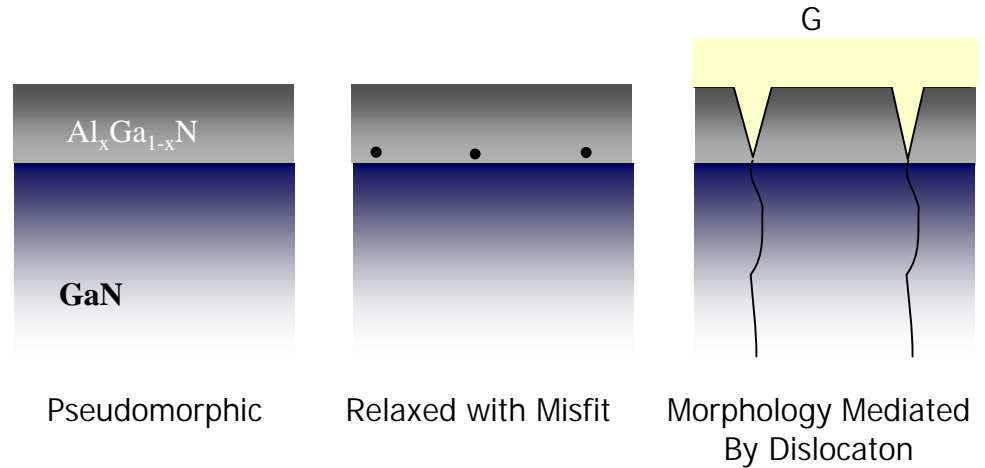
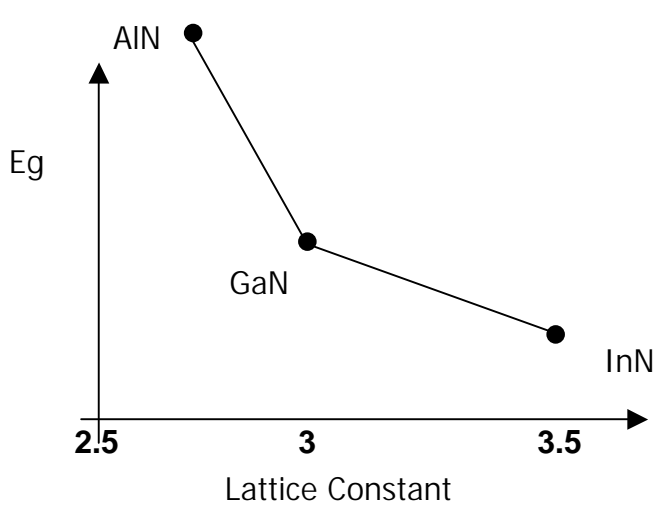
- 2DEG increases when AlN is thicker



- $I_{\max} = 950$ mA/mm
- $g_m = 200$ mS/mm

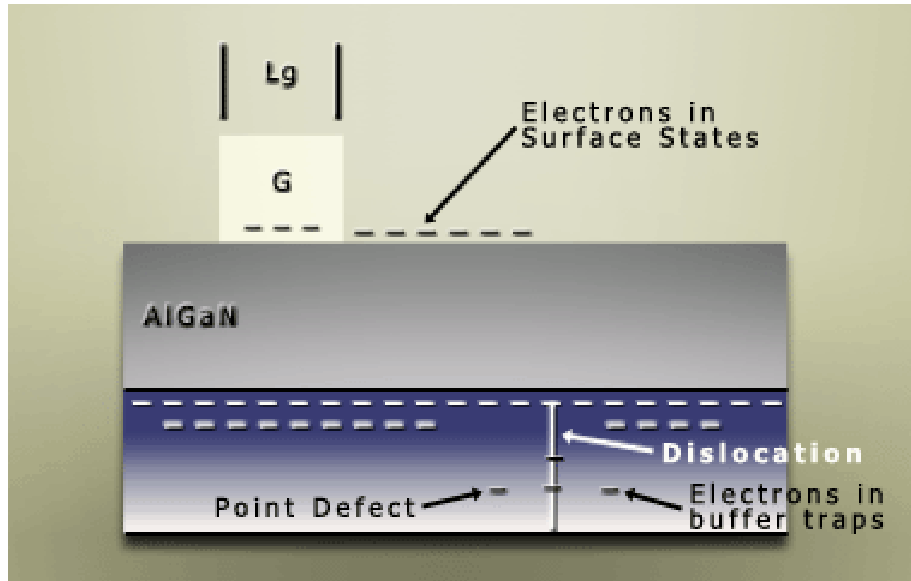


- 8.47 W/mm with a PAE of 28% @ 8GHz
- Bias: class AB at 45 V \times 160 mA/mm
- Gate dimension: $0.7 \times 150 \mu\text{m}^2$

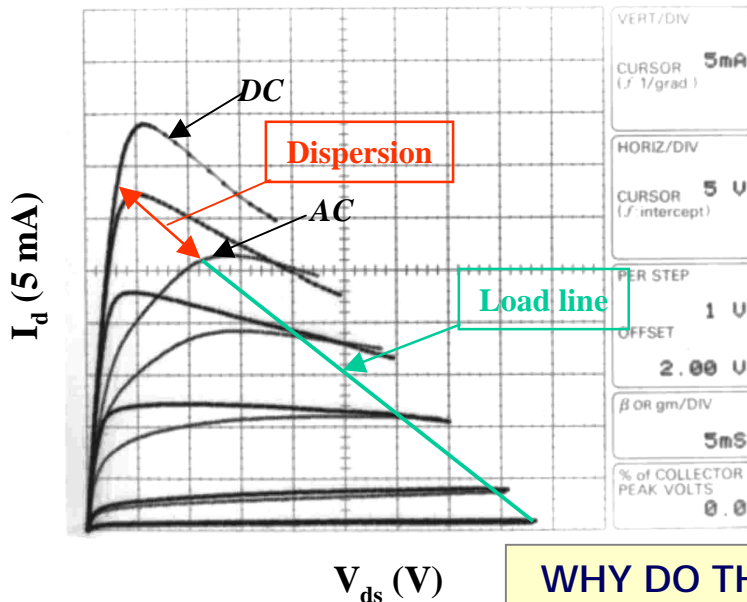


DISLOCATIONS LEAD TO PREMATURE RELAXATION OF AlGaInN AND A POTENTIAL RELIABILITY PROBLEM BECAUSE OF THE METALLIZED PITS

Minimizing Gate Length Extension

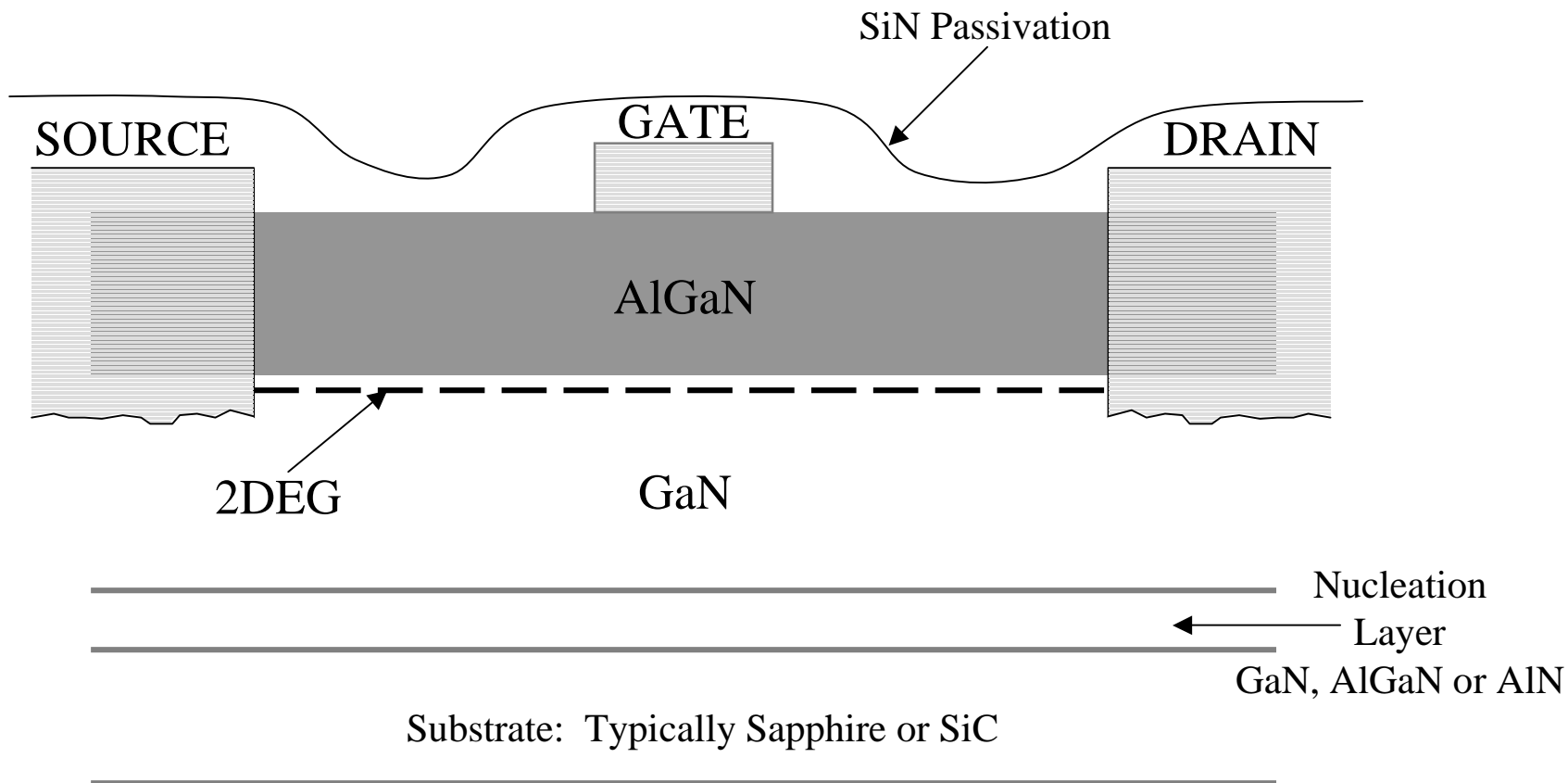


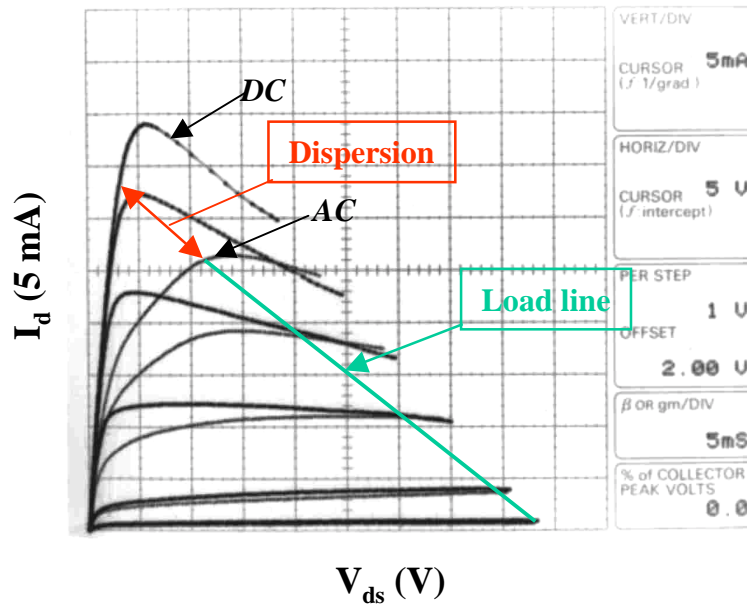
ELECTRONS IN SURFACE STATES AND/OR BUFFER TRAPS DEplete THE CHANNEL CAUSING GATE LENGTH EXTENSION

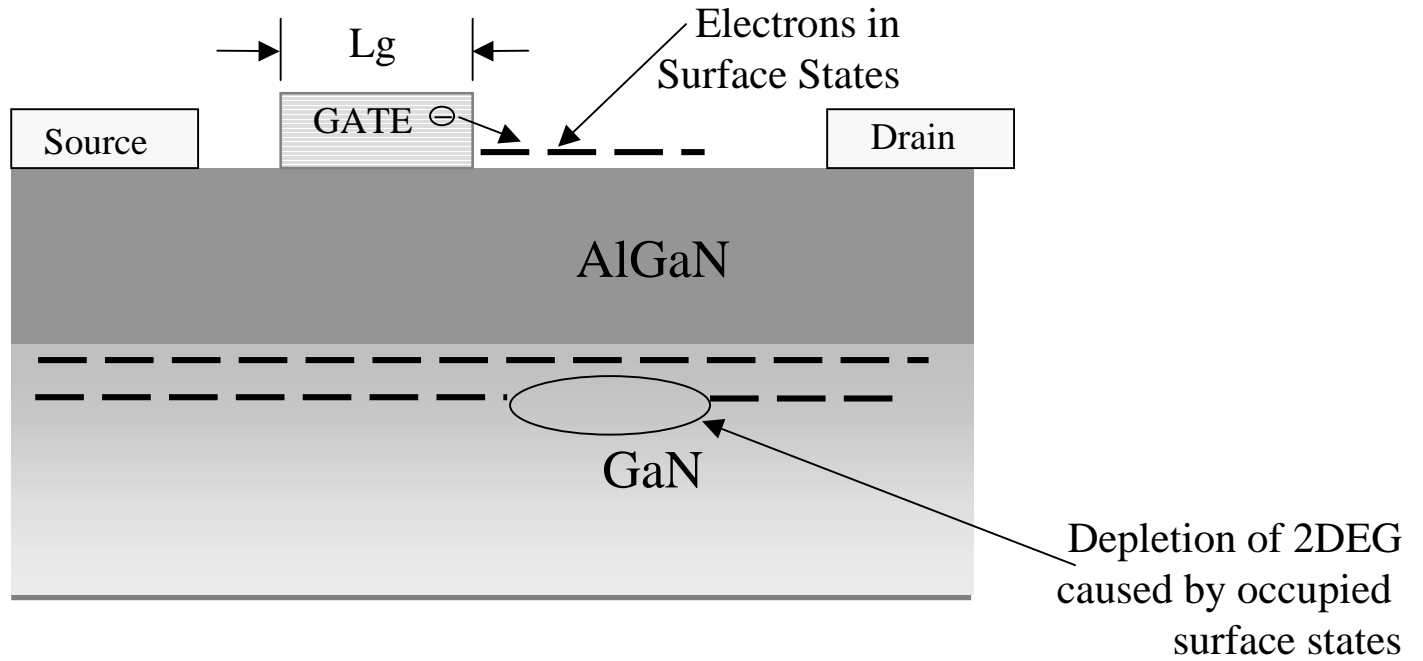


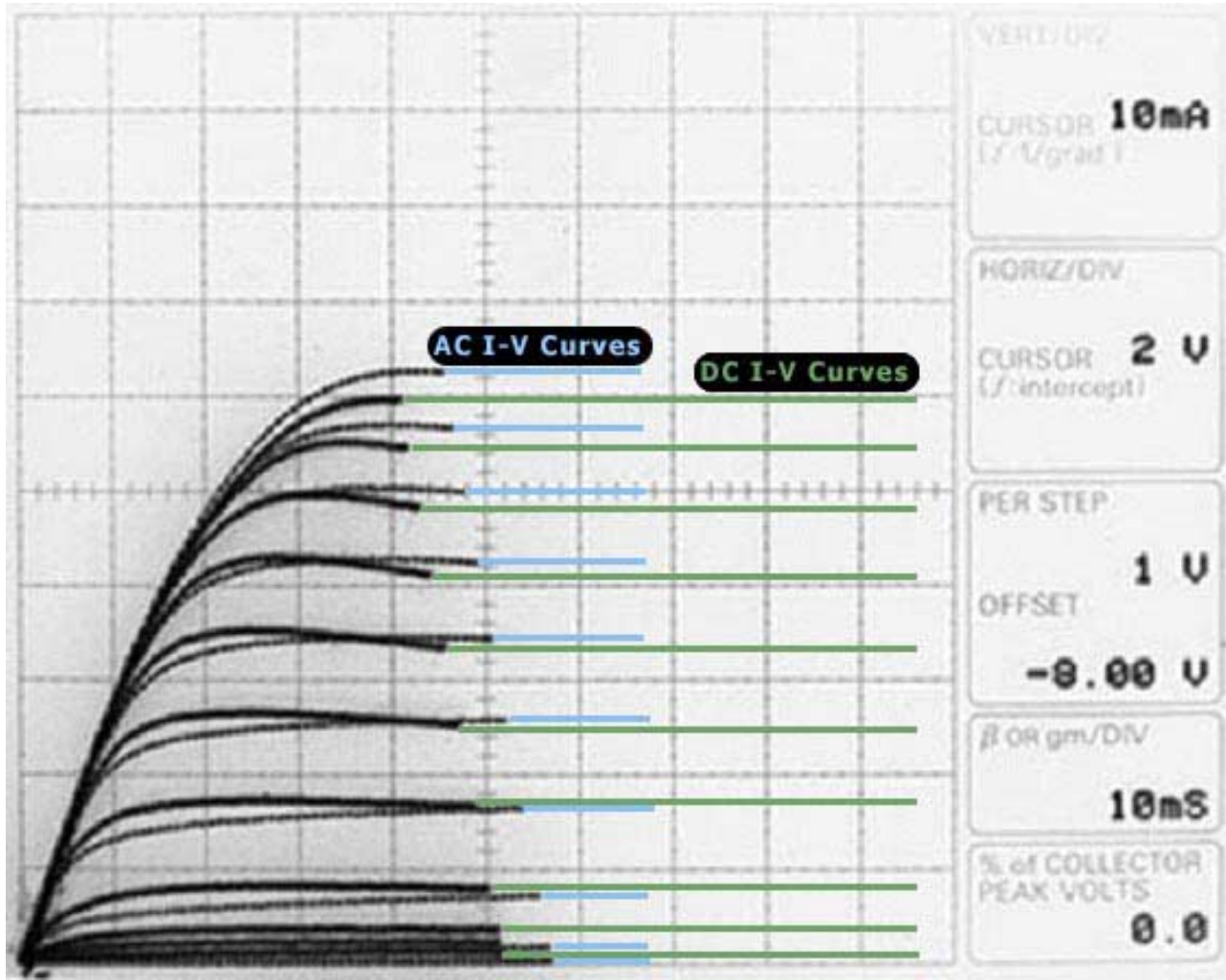
SEVERE CONSEQUENCE: DISPERSION BETWEEN SMALL SIGNAL AND LARGE SIGNAL BEHAVIOR BECAUSE OF THE LARGE TRAP TIME CONSTANTS

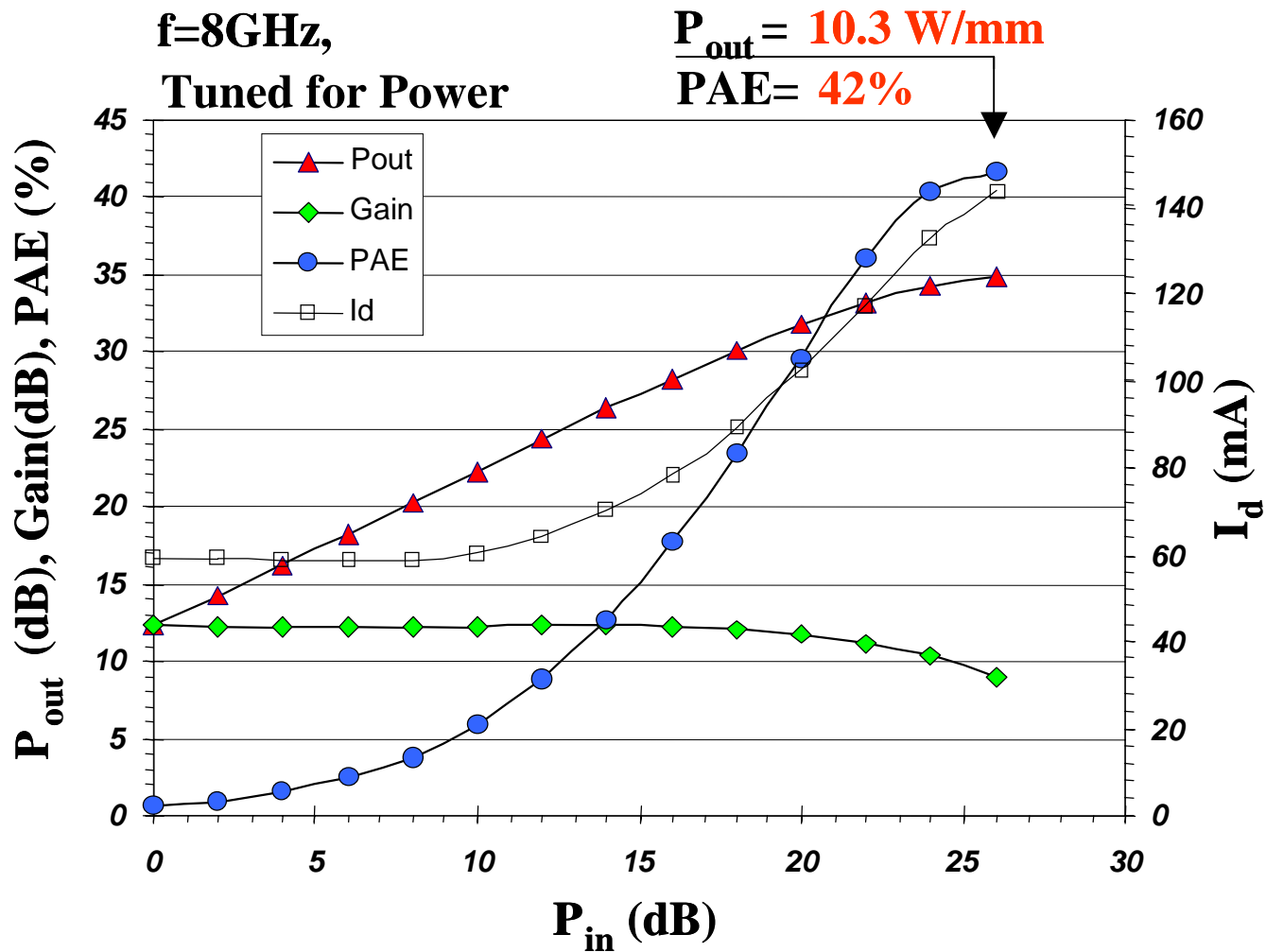
WHY DO THESE TRAPS ARISE?

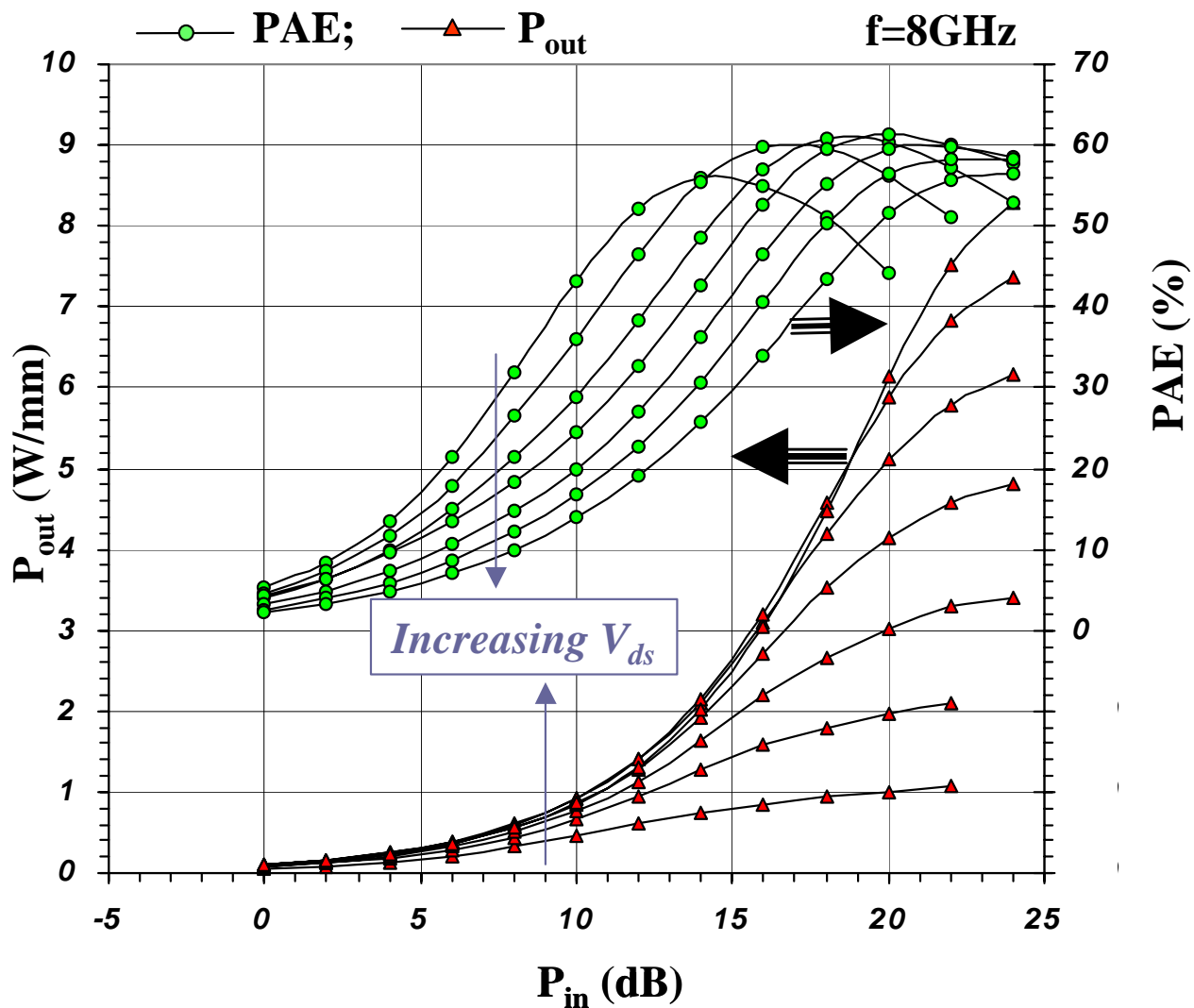


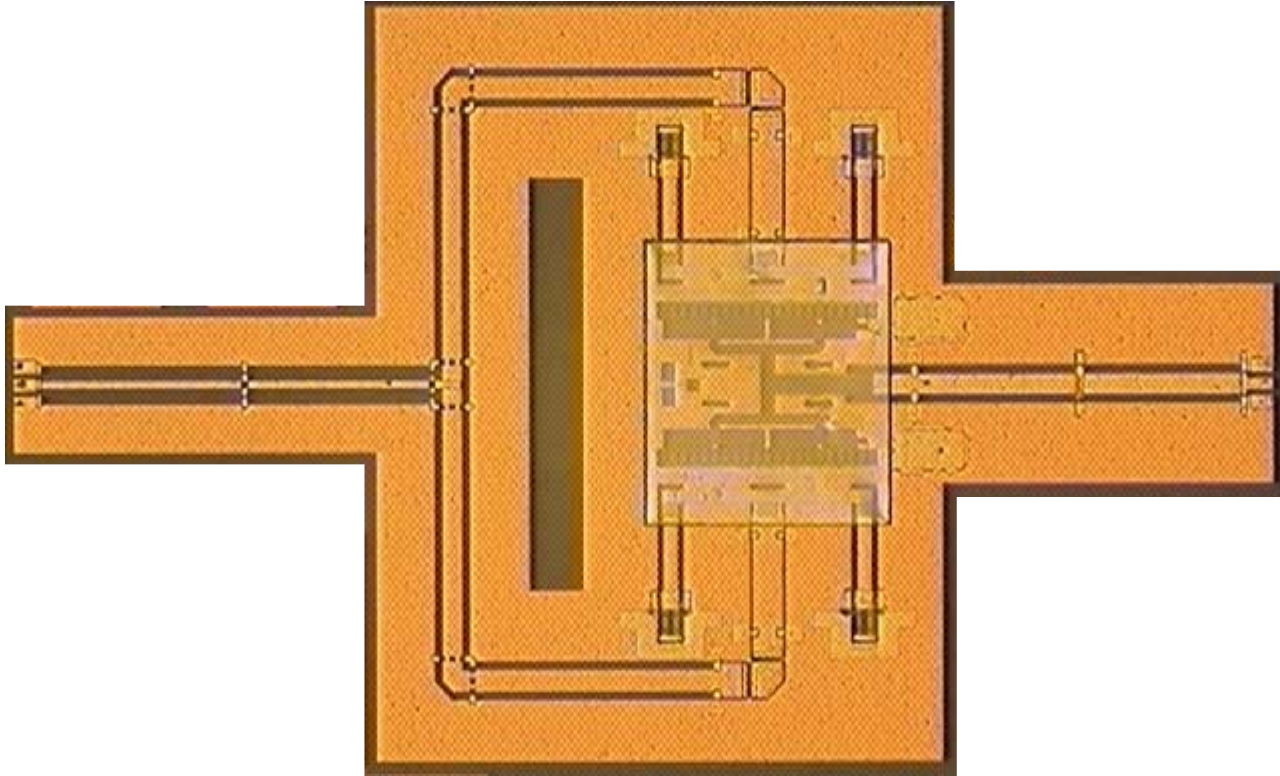




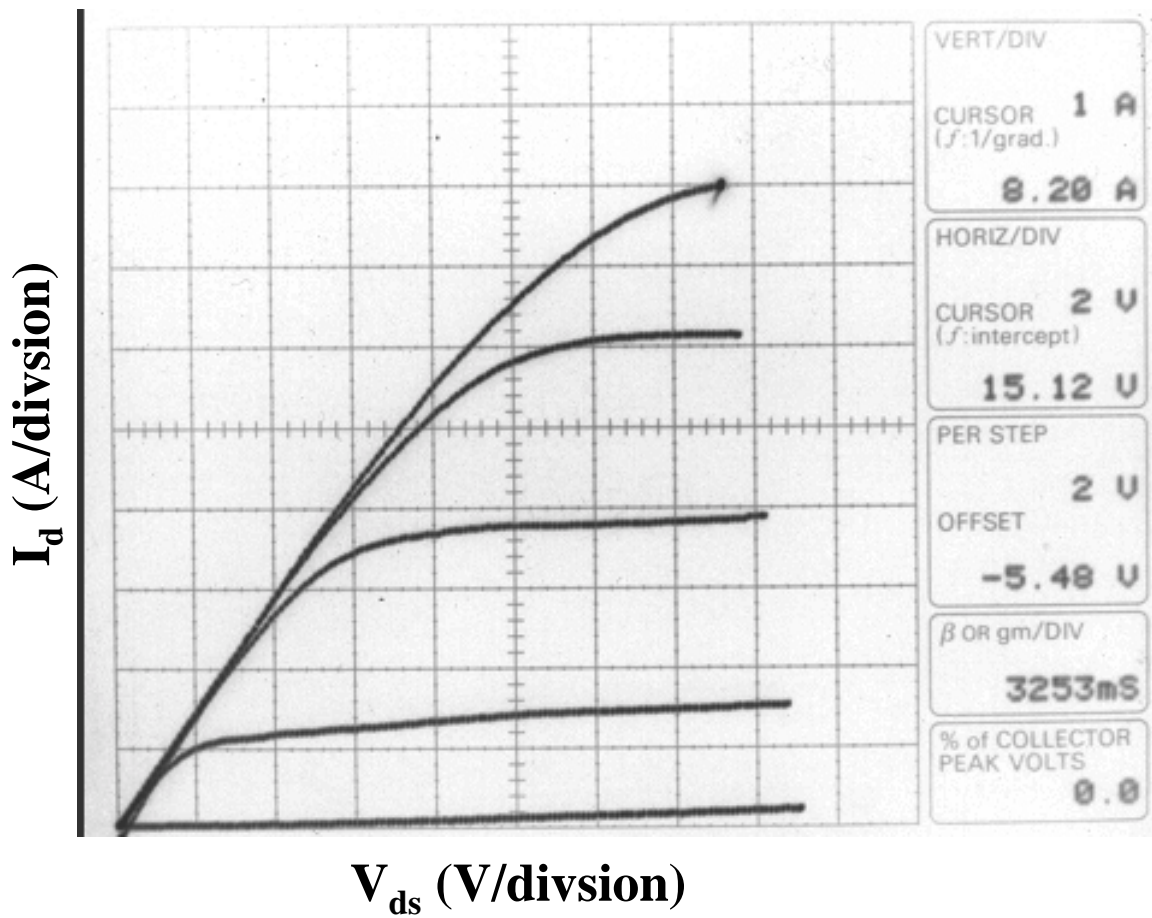


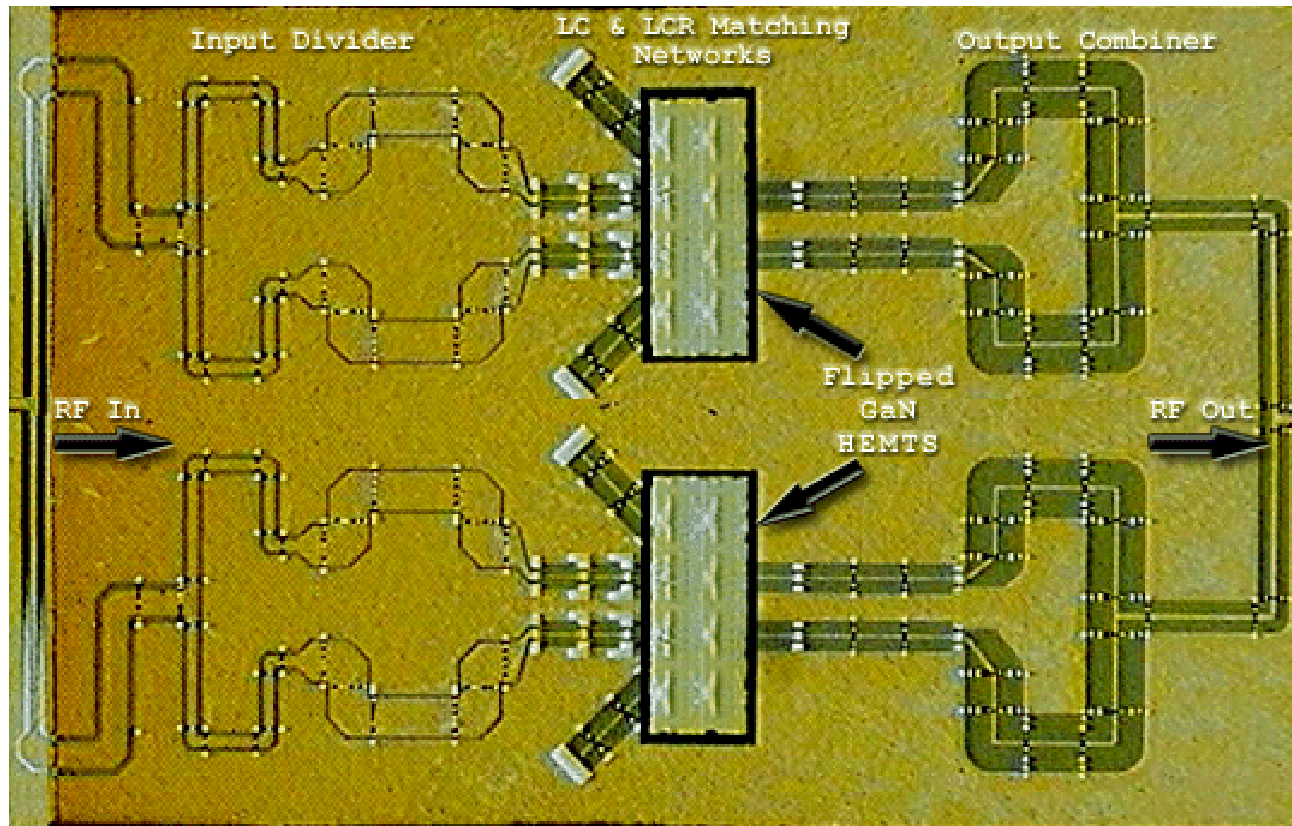


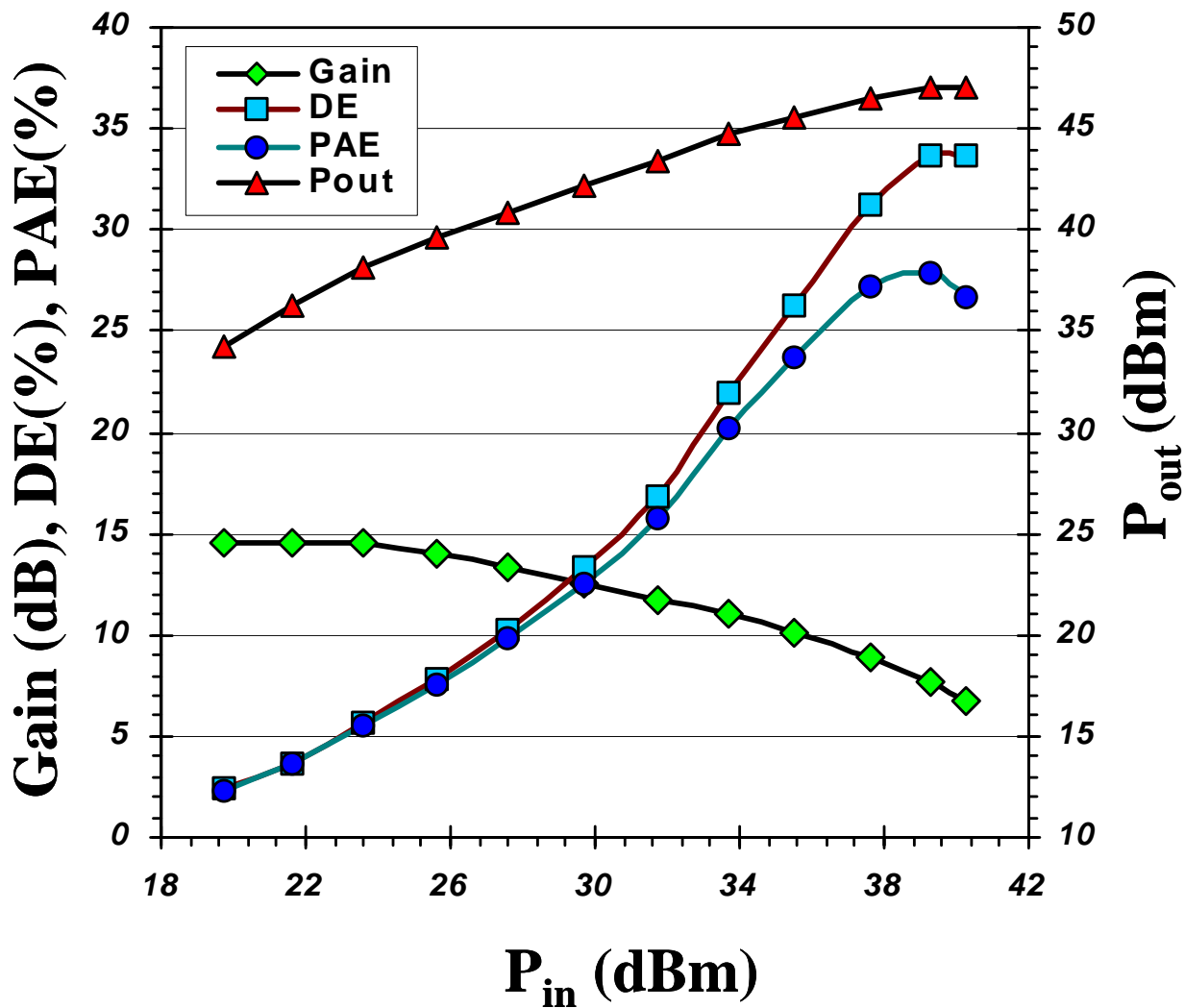


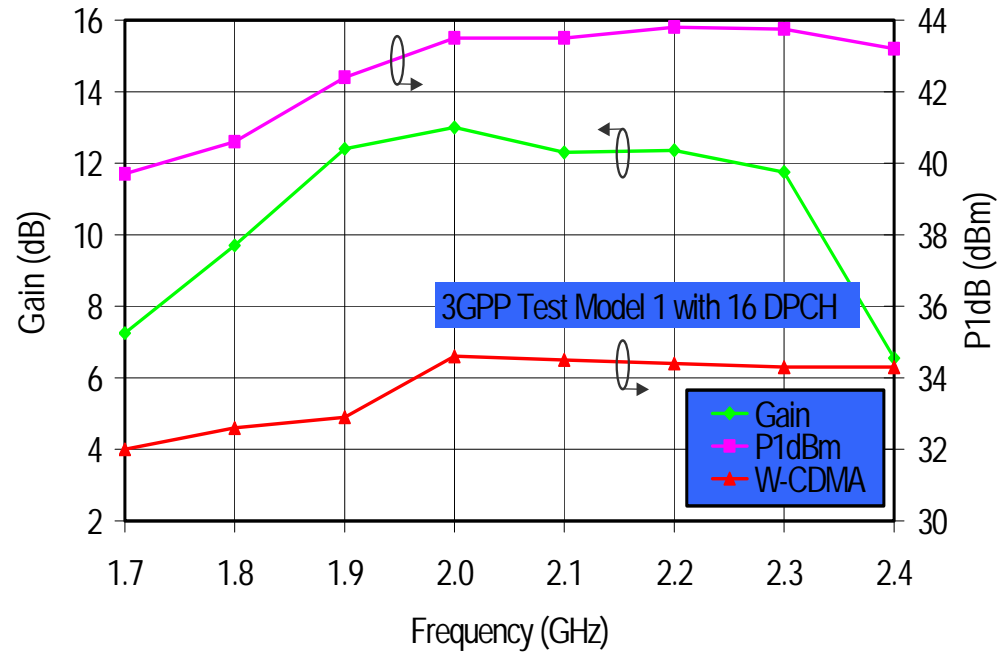
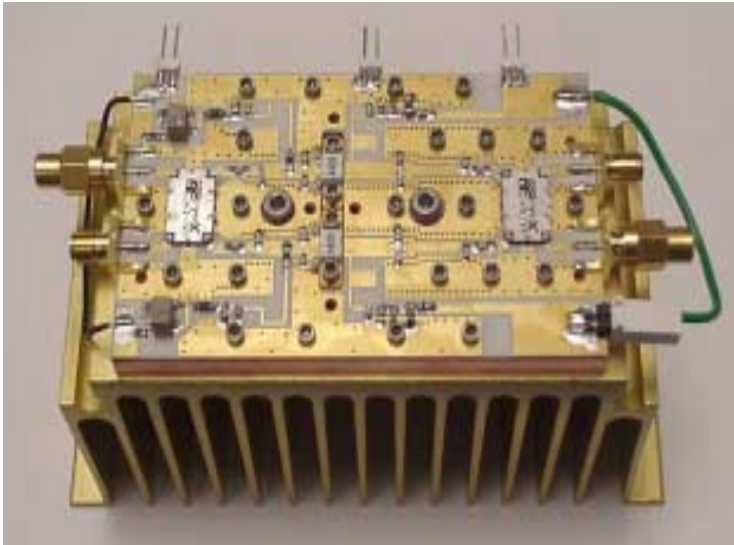


V_g start: +2V, Step: -2 V



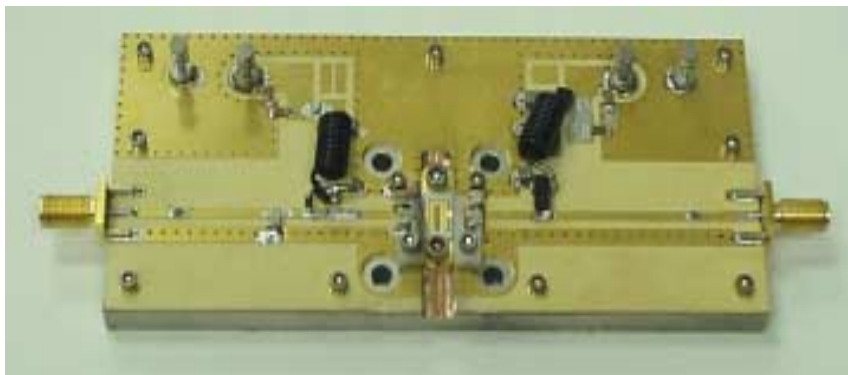




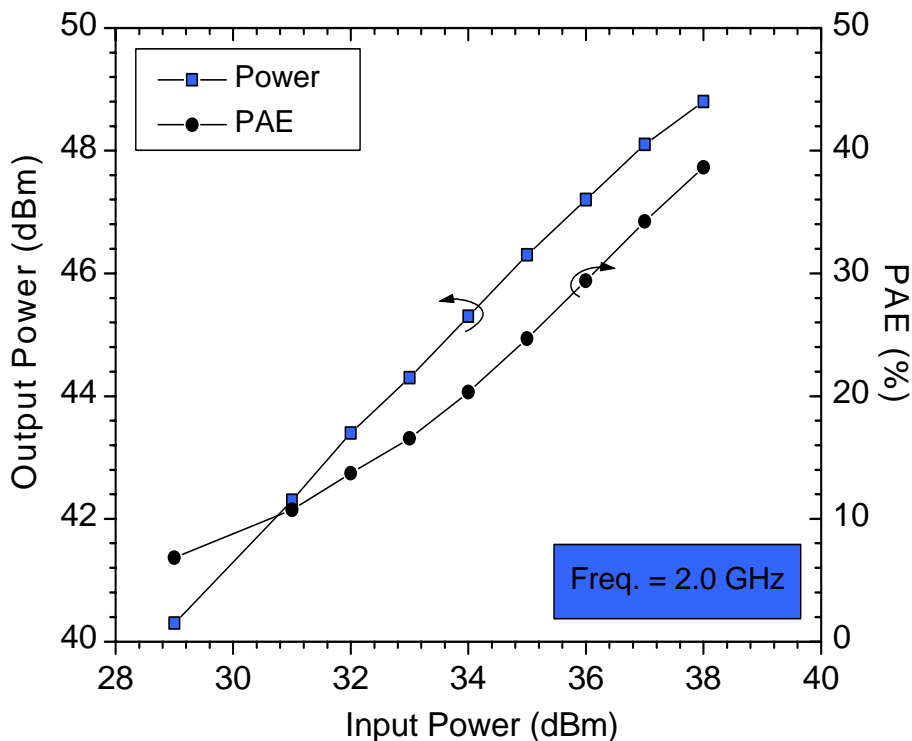


Balanced amplifier with Cree's 10-Watt commercial FETs, CRF22010

- 22 W at P_{1dB} across a 400 MHz band
- Advantage of wide bandgap transistors: power-bandwidth product greatly exceeding Si LDMOS

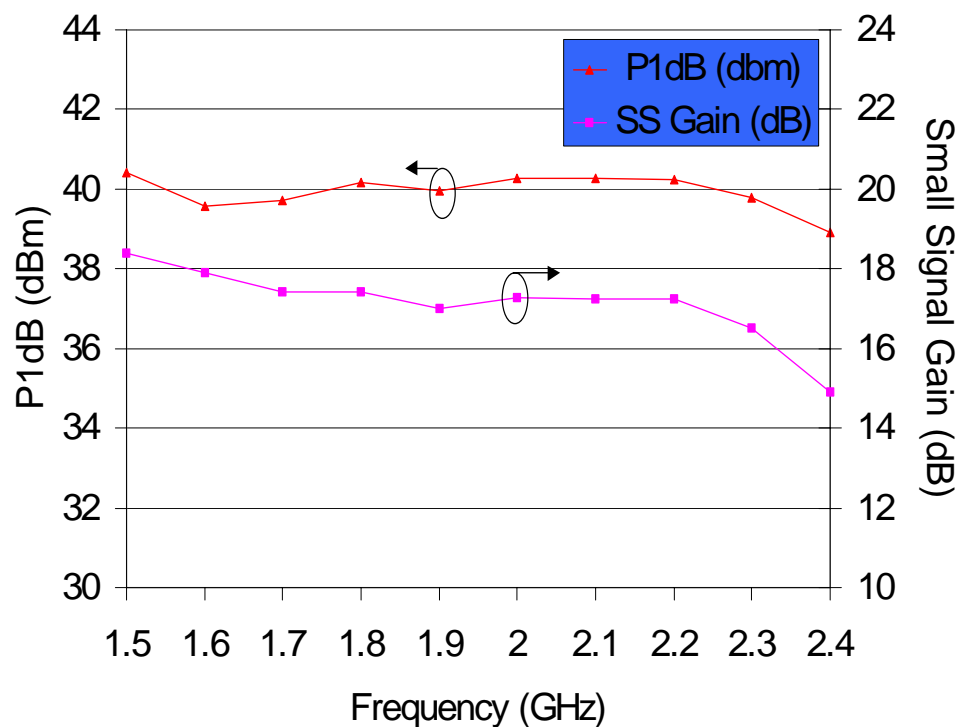
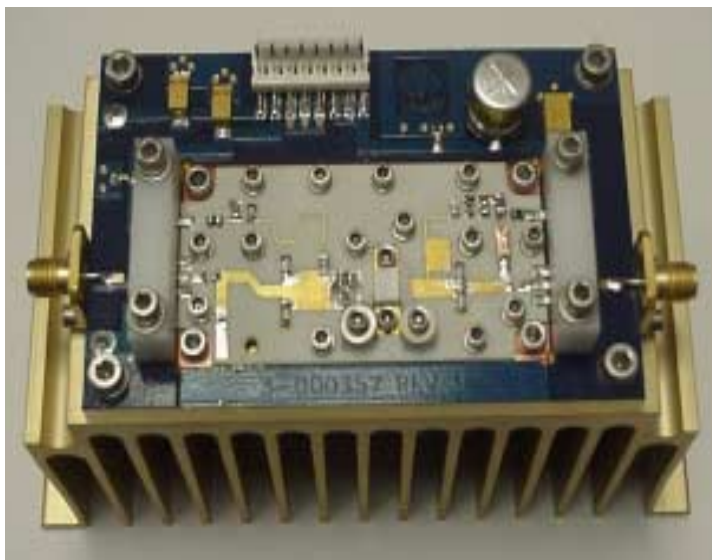


2 GHz test fixture for 60 W MESFET development

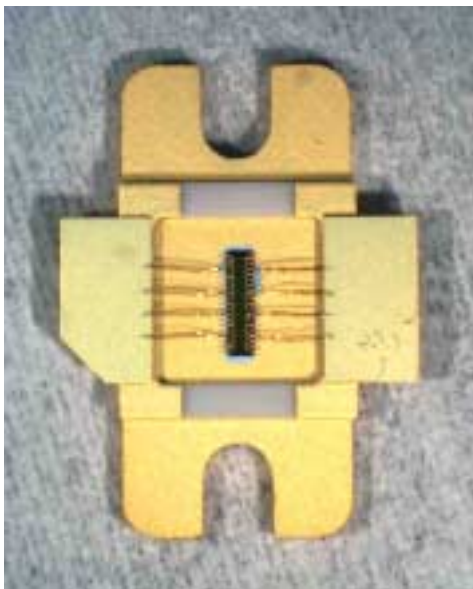


- 75 W CW, 11 dB gain demonstrated from a single SiC MESFET
- Currently being optimized for a 60-Watt Class A MESFET product, targeted for release by the end of the year

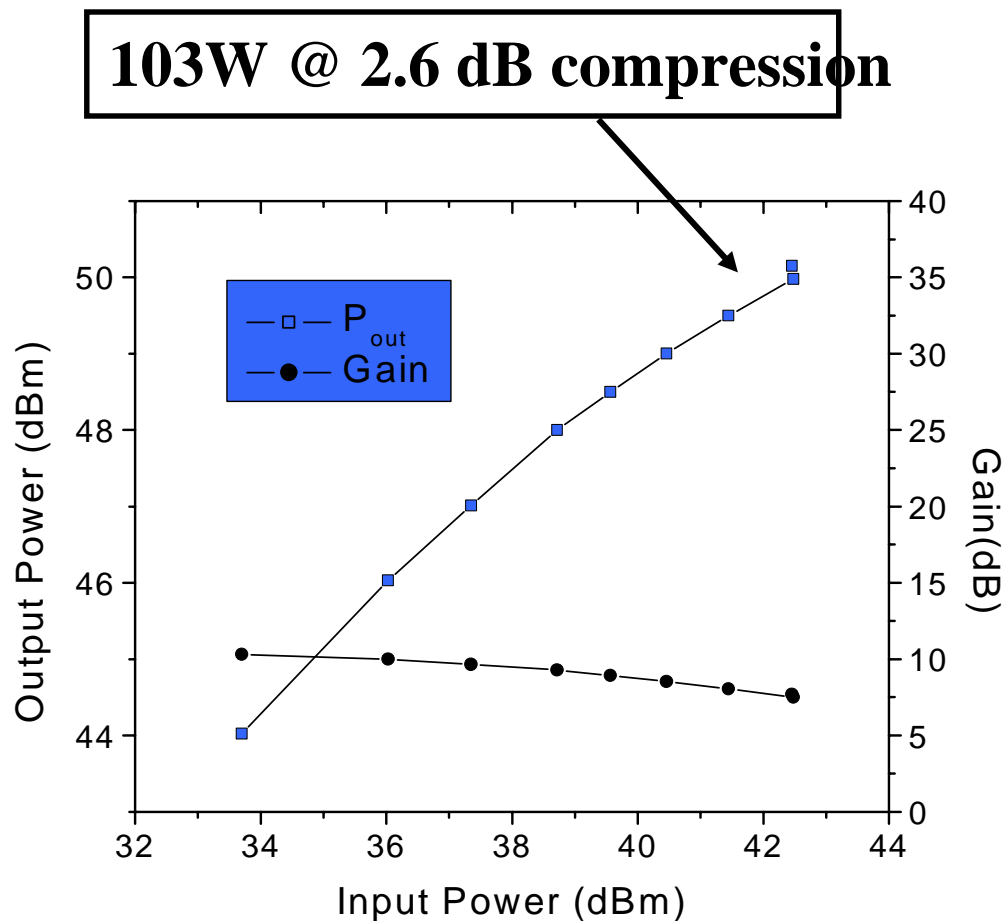
10-Watt Broadband GaN HEMT Amplifier



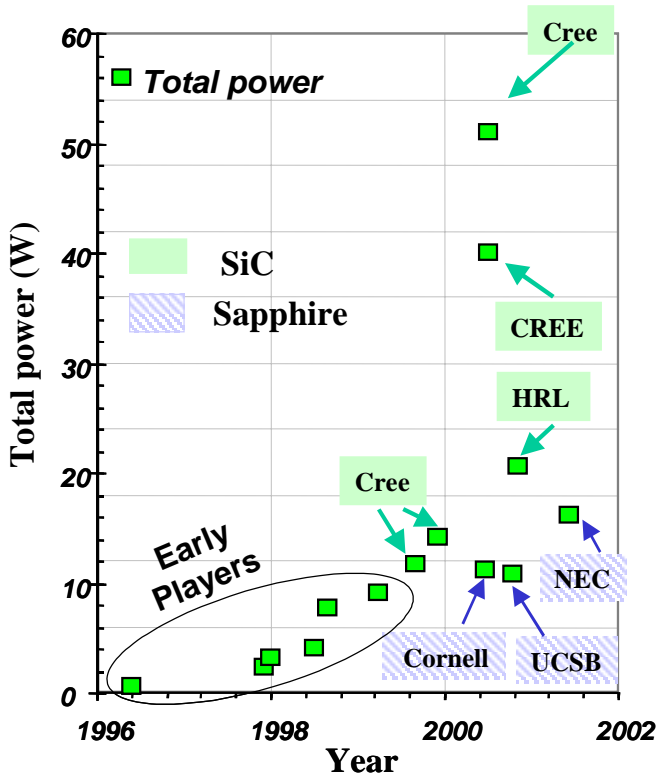
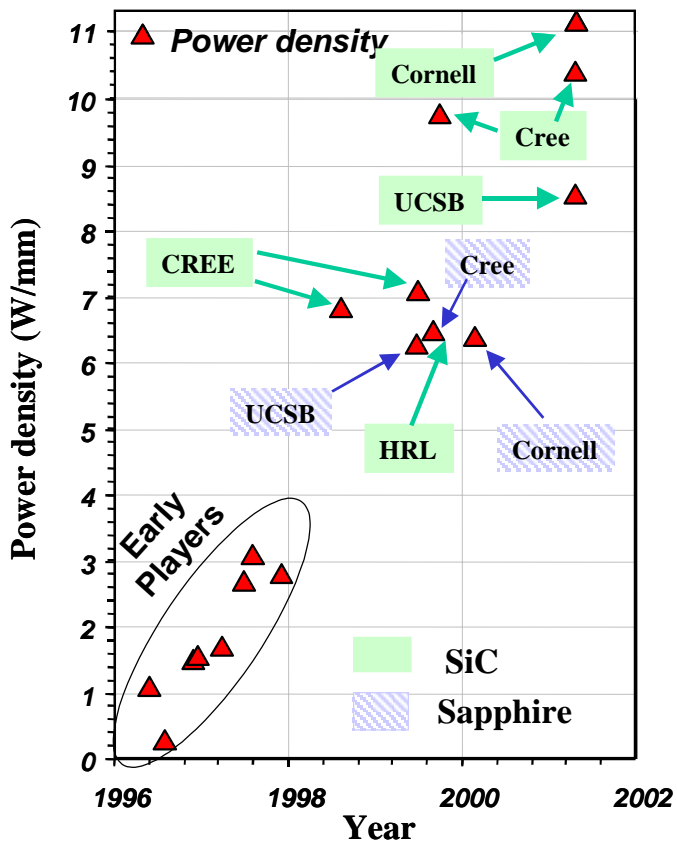
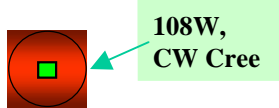
- 11 W at P_{1dB} across the 1.8 - 2.2 GHz band
- 17 dB gain with only ± 0.15 dB ripple



- Freq. = 2 GHz
- $V_{DS} = 52V$
- Peak Drain Eff. = 54%
- 108 W achieved at P_{3dB}



- Commercially-available SiC MESFETs
 - 22 W broadband amplifier demonstrates power-bandwidth advantage of this technology
 - 60 W Class A MESFET targeted for release by end of year
- GaN HEMTS on SiC Substrates
 - excellent broadband and high power performance demonstrated
 - emphasis of development is now on reproducibility and reliability
- SiC-based MMIC process developed for both technologies
- 3-inch SI substrates will enable cost-competitive manufacturing



- Includes **ALL LEADING** players in the field
- CREE = **Cree Lighting** + **Cree-Durham**

Part II

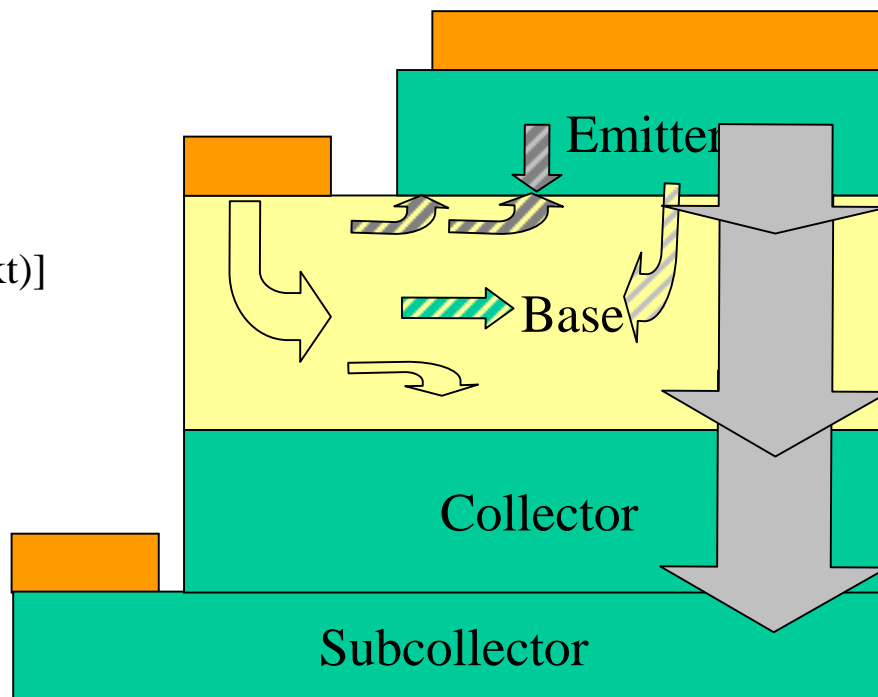
High Voltage Operation (> 330 V) of AlGa_N/Ga_N HBTs

- Injection

- $\gamma \Rightarrow 1$

- $n \Rightarrow 1$

- [$I = I_0 \exp(qv/nkt)$]



Transport

-- $\alpha \Rightarrow 1$

Collection

- $C_{bc} \Rightarrow 0$

- $v \Rightarrow v_{sat}$ [2×10^7 cm/s] (*Kolnik et. al.*)

- $V_{br} \Rightarrow E_{crit} W_C$ [$E_{crit} \sim 2$ MV/cm] (*Bhaskar and Shur.*)

Output Conductance

- $\Delta I_C / \Delta V_{CE} \Rightarrow 0$

- ($\Delta W_B / \Delta V_{CE} \Rightarrow 0$)

UCSB Hurdles with GaN bipolar transistors

- Lack of low damage etch to reveal base

- Leaky E/B junction
- Bad base contact
- No etch stop

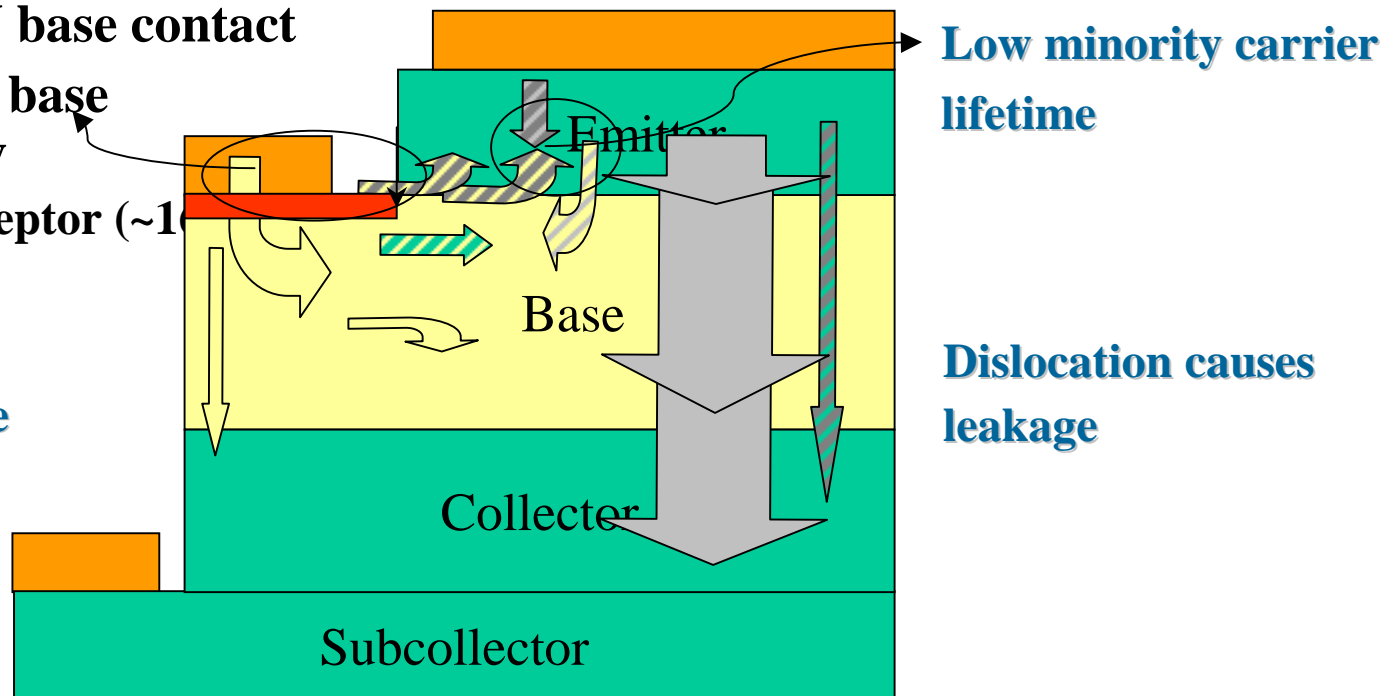
- High R_B

- Poor p-GaN base contact
- Low p-GaN base conductivity

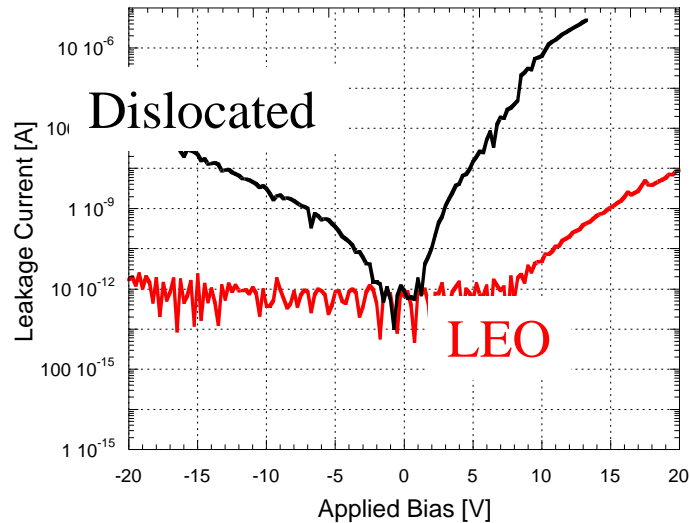
- Deep acceptor (~ 1)

Surface leakage due to etch damage

- Hard to control junction placement in MOCVD due to memory effect of p-dopant Mg



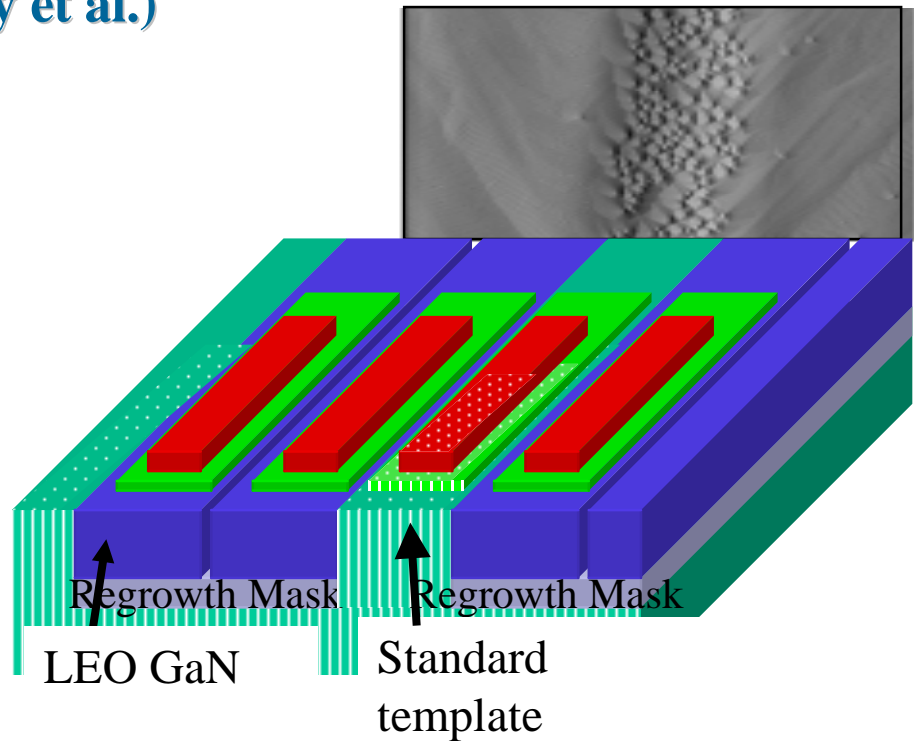
LEO used to investigate leakage of devices without dislocations. (Lee McCarthy et al.)



Leakage from Collector to Emitter, Wing vs Window

- Results:** LEO device demonstrated
- Reduction in Leakage
 - Stable operation past 20V
 - Gain unchanged
 - Devices on dislocated material also functional

AFM scan of wing vs Window on LEO GaN

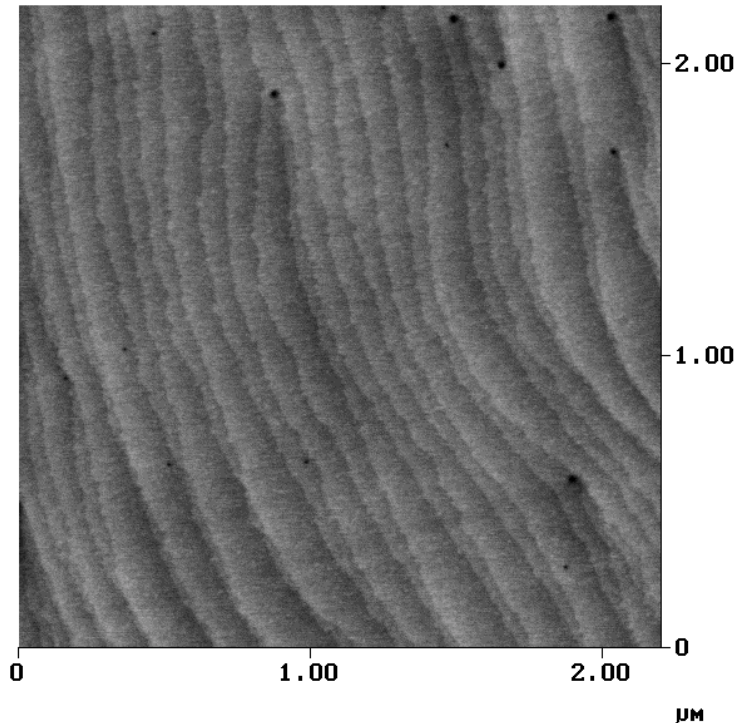


Explanation

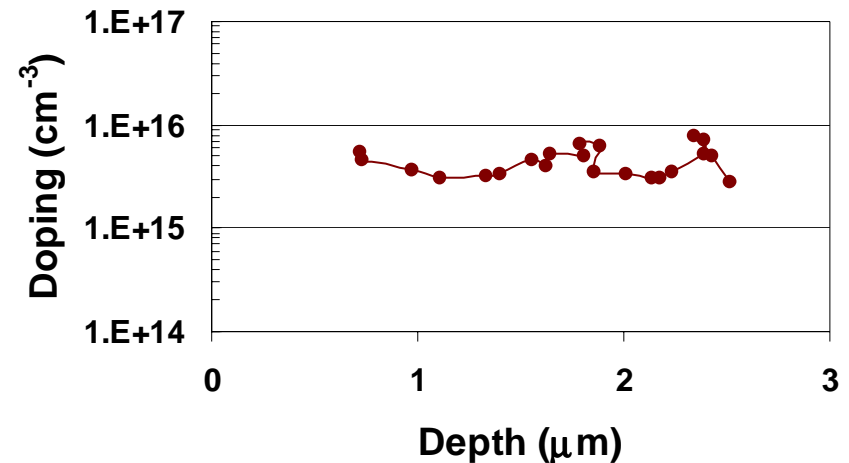
- Thick substrate sufficiently reduces dislocations to prevent C/E short in window region
- Gain (τ_e) not currently limited by dislocation density

UCSB Strategy: Thick Collector

- Decent dislocation density
 - High quality MOCVD templates achieved
Dislocation density $\sim 5e8 \text{ cm}^{-2}$
- Low background doping
 - $N_D < 1e16 \text{ cm}^{-3}$ (Assuming uniform doping N_D and $E_{\text{critical}} = 2 \text{ MV/cm}$, requires $10 \mu\text{m}$ to achieve 1 KV breakdown voltage.)



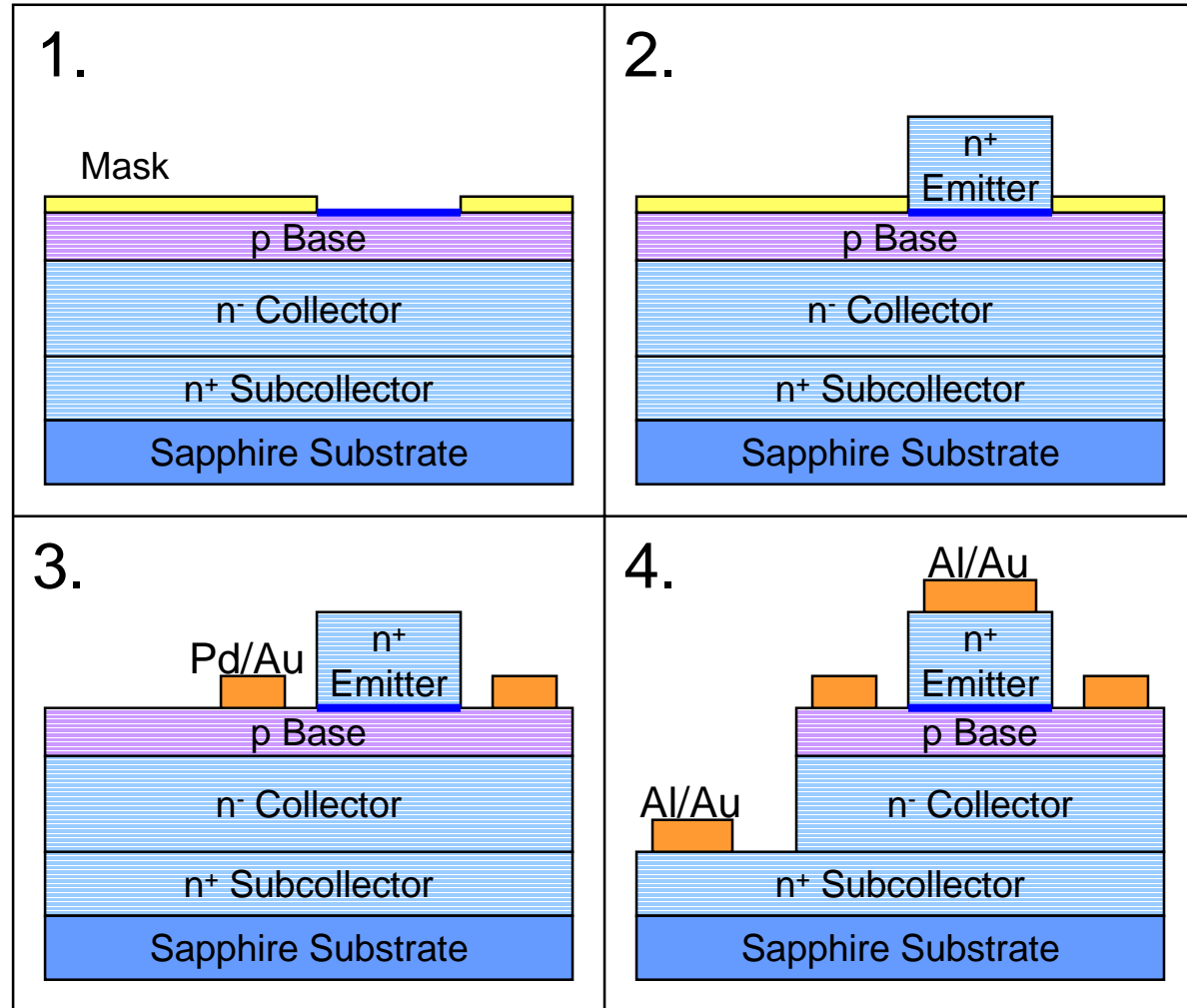
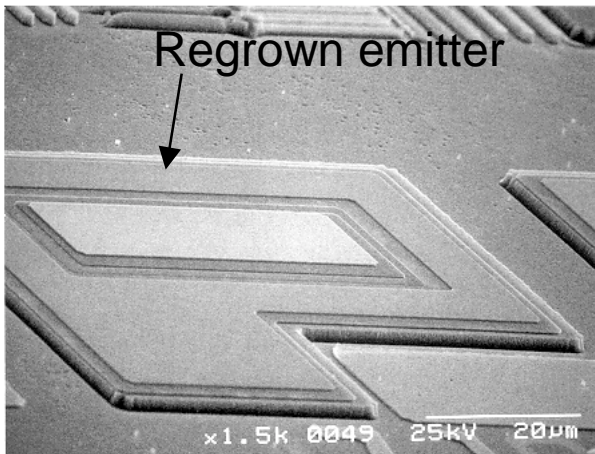
Doping vs. Depth
(010704GA, 8 μm collector)



UCSB Emitter Regrowth Process Flow

- **Selectively grow MOCVD emitter on base-collector structures.**

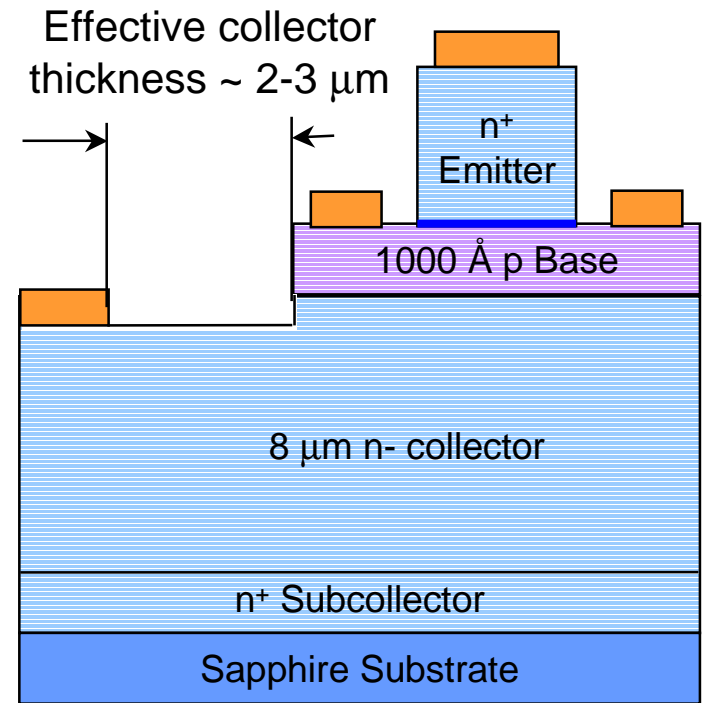
1. Pattern regrowth mask
2. Regrow emitter layer by MOCVD
3. Remove mask and contact base and etch to collector
4. Contact collector, emitter



UCSB Device structure

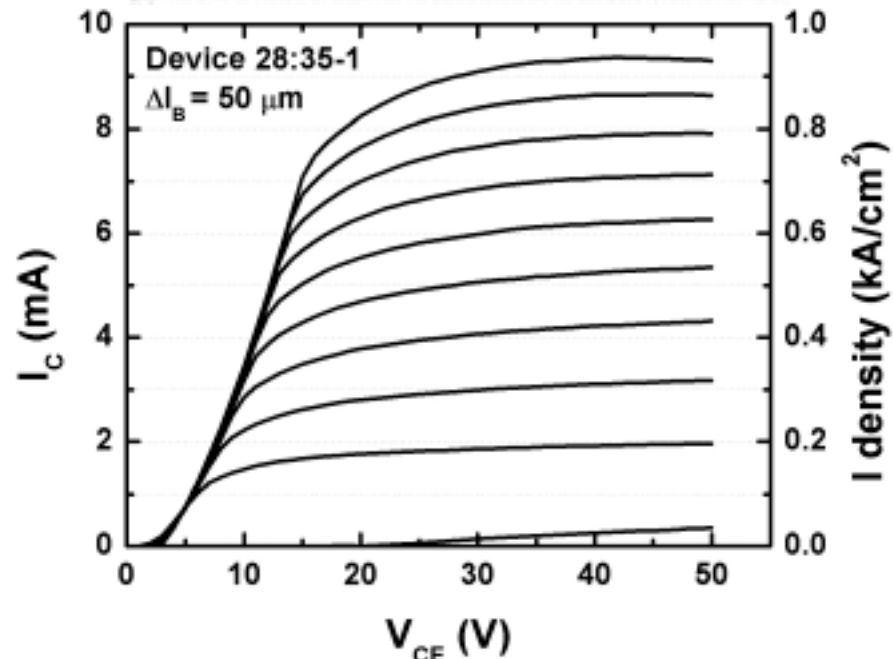
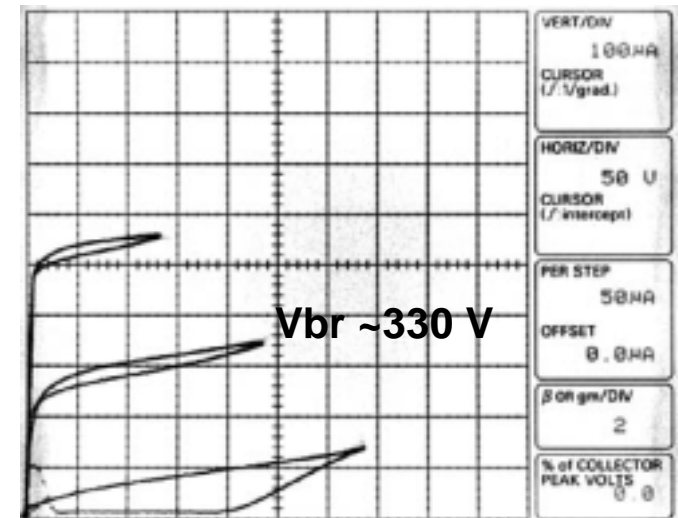
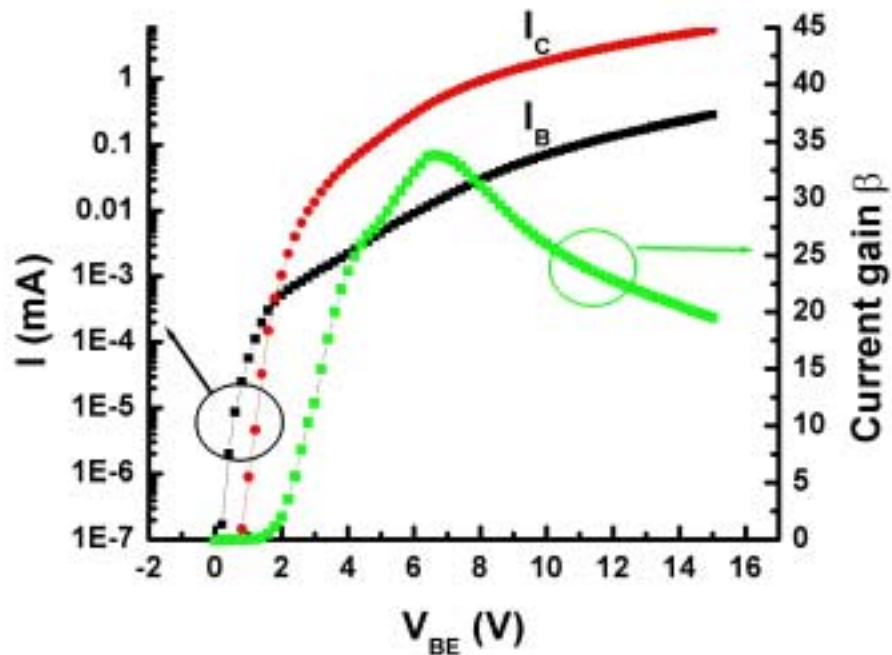
- Utilization of uid GaN spacer and grading layer
 - HBTs with high emitter injection efficiency
- Etch damage and current mask layout limits V_{br}

4 nm GaN:Si ($1e18 \text{ cm}^{-3}$) contact
4 nm $\text{Al}_{0.05} \text{ GaN} \rightarrow \text{GaN:Si}$ ($1e18 \text{ cm}^{-3}$) grading
105 nm $\text{Al}_{0.05} \text{ GaN:Si}$ ($1e18 \text{ cm}^{-3}$) emitter
8 nm $\text{GaN} \rightarrow \text{Al}_{0.05} \text{ GaN}$ ($?3e18 \text{ cm}^{-3}$) grading
8 nm uid GaN spacer
100 nm GaN:Mg ($2e19 \text{ cm}^{-3}$) base
8 μm uid GaN ($4e15 \text{ cm}^{-3}$) collector
2 μm GaN:Si ($1e18 \text{ cm}^{-3}$) subcollector
Sapphire



UCSB HBT with 8 mm GaN collector

- Current gain (β) > 20
- Common emitter operation > 300 V
- Non-passivated
- Base thickness 1000 Å
- $\text{Al}_{0.05}\text{GaN}$ emitter



- Conclusion
 - In selective emitter regrowth, a sharp Mg profile, ~ 40 nm/decade, enables the precise junction placement
 - Improvement of regrown-emitter/base diodes
 - Demonstration of high V_{br} (> 300 V) with high β (DC common emitter operation up to 35)